

SYNTHESES AND INVESTIGATION OF ELECTROCHEMICAL PROPERTIES OF  
SELENOPHENE AND BENZOTRIAZOLE BEARING COPOLYMERS

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HANDE ÜNAY

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**SYNTHESES AND INVESTIGATION OF ELECTROCHEMICAL  
PROPERTIES OF SELENOPHENE AND BENZOTRIAZOLE CONTAINING  
COPOLYMERS**

submitted by **HANDE ÜNAY** in partial fulfillment of the requirements for the degree  
of **Master of Science in Polymer Science and Technology Department, Middle East  
Technical University** by,

Prof. Dr. Canan Özgen  
Dean, Graduate School of **Natural and Applied Sciences**

\_\_\_\_\_

Prof. Dr. Teoman Tinçer  
Head of Department, **Polymer Science and Technology**

\_\_\_\_\_

Assoc. Prof. Dr. Ali Çırpan  
Supervisor, **Chemistry Dept., METU**

\_\_\_\_\_

Assoc. Prof. Dr. H. Emrah Ünalın  
Co-supervisor, **Metallurgical and Materials Eng. Dept., METU**

\_\_\_\_\_

**Examining Committee Members:**

Prof. Dr. Levent Toppare  
Chemistry Dept., METU

\_\_\_\_\_

Assoc. Prof. Dr. Ali Çırpan  
Chemistry Dept., METU

\_\_\_\_\_

Asisst. Prof. Dr. İrem Erel Göktepe  
Chemistry Dept., METU

\_\_\_\_\_

Assoc. Prof. Dr. Yasemin Arslan Udum  
Adv. Tech. Dept., Gazi Univ.

\_\_\_\_\_

Dr. Görkem Günbaş  
Chemistry Dept., METU

\_\_\_\_\_

**Date:** 05/09/2014

**I hereby declare that all information in this document has been obtained and presented in accordance with academic rules and ethical conduct. I also declare that, as required by these rules and conduct, I have fully cited and referenced all material and results that are not original to this work.**

Name, Last name: Hande Ünay

Signature:

## ABSTRACT

# SYNTHESES AND INVESTIGATION OF ELECTROCHEMICAL PROPERTIES OF SELENOPHENE AND BENZOTRIAZOLE BEARING COPOLYMERS

Ünay, Hande

M. S., Polymer Science and Technology Department

Supervisor : Assoc.Prof. Dr. Ali Çırpan

Co-Supervisor : Assoc.Prof. Dr. Emrah Ünalın

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Polymers containing selenophene as a  $\pi$  bridge; poly(2-(2-octyldodecyl)-4-(selenophen-2-yl)-7-(5-(thiophen-2-yl)selenophen-2-yl)-2H-benzo[d][1,2,3]triazole) (**P-SBTTh**) and poly(2-(2-octyldodecyl)-4-(5-phenylselenophen-2-yl)-7-(selenophen-2-yl)-2H-benzo[d][1,2,3]triazole (**P-SBTPh**) were synthesized via Stille and Suzuki coupling, respectively. Optical and electrochemical properties were investigated and comparisons with their thiophene analogs were done. Selenophene substitution resulted in low band gap and red shifted absorption, high switching times (less than 1s) with reasonable optical contrast. Although both polymers showed moderate stability during chronoamperometry studies, **P-SBTPh** showed high stability in NIR region with fast

switching times. Switching time of **P-SBTPh** was 0.2 second that is the fastest switching time compared to all benzotriazole-based polymers.

**Keywords:** Selenofen, Low Band Gap, Fast Switching Time, Benzotriazole

## ÖZ

### SELENOFEN VE BENZOTİAZOL İÇEREN KOPOLİMERLERİN SENTEZİ VE ELEKTROKİMYASAL ÖZELLİKLERİNİN İNCELENMESİ

Ünay, Hande

Yüksek Lisans, Polimer Bilimi ve Teknolojisi Bölümü

Tez Yöneticisi : Doç.Dr. Ali Çırpan

Ortak Tez Yöneticisi : Doç.Dr. Emrah Ünalın

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Bu çalışmada,  $\pi$ -köprüsü olarak selenofen içeren poli(2-(2-oktildodesil)-4-(selenofen-2-yl)-7-(5-(tiyofen-2-il)selenofen-2-il)-2H-benzo[d][1,2,3]triazol) (**P-SBTTh**) ve poli(2-(2-oktildodesil) -4 (5-fenilselenofen-2-il) -7 (selenofen-2-il)-2H-benzo [d] [1,2,3] triazol (**P-SBTPh**) sırasıyla Still eve Suzuki kenetlenme reaksiyonları ile sentezlenmiştir. Optik ve elektrokimyasal özellikleri incelenmiş ve tiyofen analoglarıyla bu özelliklerin karşılaştırmaları yapılmıştır. Selenofen eklenmesi düşük bant aralığına ve absorpsiyonun kırmızı alana kaymasına ve kayda değer optik kontrast ile hızlı anahtarlama zamanına (1sn'den daha az) neden olmuştur. Her iki polimerde kronoamperometri çalışmalarında düşük kararlılık gösterse de, **P-SBTPh** yakın

kızılötesi alanda yüksek kararlılık ve hızlı anahtarlama zamanına sahip olduğu gözlenmiştir.**P-SBTPh** 0.2 sn anahtarlama zamanı ile benzotriazol temelli tüm polimerler içerisinde en hızlı anahtarlama zamanına sahip polimerdir.

**Anahtar kelimeler:** Selenofen, Düşük Bant Aralığı, Yüksek Anahtarlama Zamanı, Benzotriazol.



*To my precious family*

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## LIST OF ABBREVIATIONS

<b>CP</b>	Conjugated Polymer
<b>PA</b>	Polyacetylene
<b>PT</b>	Polythiophene
<b>P3AT</b>	Poly (3-alkyl)thiophene
<b>PPy</b>	Polypyrrole
<b>PEDOT</b>	Polyethylenedioxythiophene
<b>PINT</b>	Polyisothianaphene
<b>PPP</b>	Poly-paraphenylene
<b>PPS</b>	Poly-paraphenylenesulfide
<b>PANI</b>	Polyaniline
<b>PPV</b>	Poly-paraphenylenevinylene
<b>HOMO</b>	Highest Occupied Molecular Orbital
<b>LUMO</b>	Lowest Unoccupied Molecular Orbital
<b>P-SBTTh</b>	Poly(2-(2-octyldodecyl)-4-(selenophen-2-yl)-7-(5-(thiophen-2-yl)selenophen-2-yl)-2H-benzo[d][1,2,3]triazole)
<b>P-SBTPh</b>	Poly(2-(2-octyldodecyl)-4-(5-phenylselenophen-2-yl)-7-(selenophen-2-yl)-2H-benzo[d][1,2,3]triazole)
<b>IR</b>	Infrared
<b>VB</b>	Valence Band
<b>CB</b>	Conduction Band
<b>BLA</b>	Bond Length Alternation
<b>D</b>	Donor

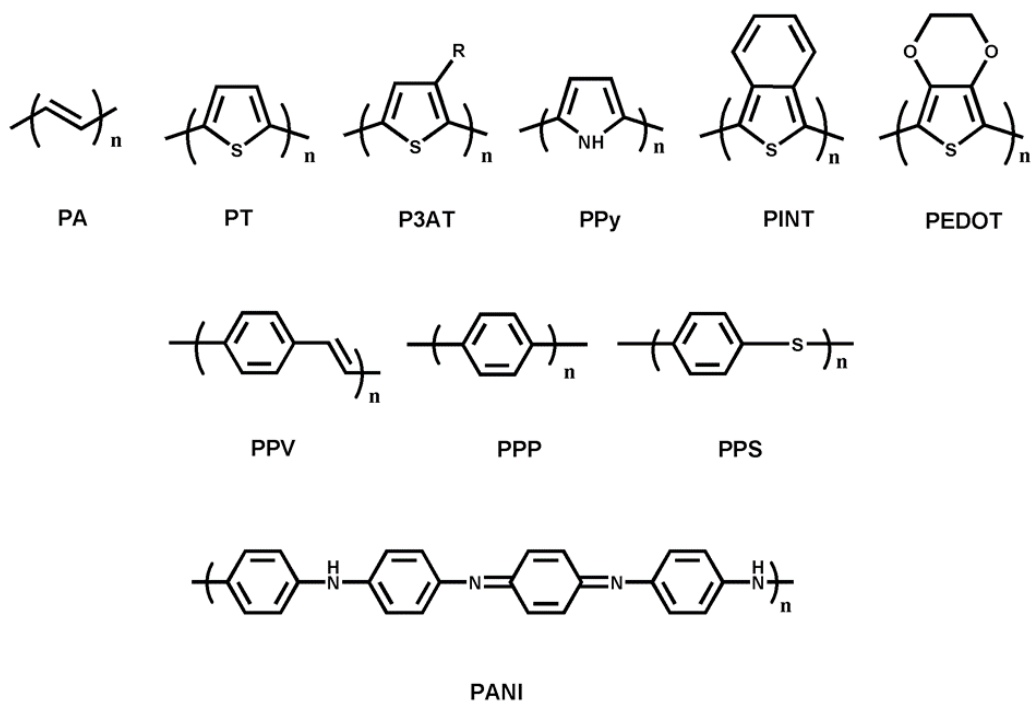
<b>A</b>	Acceptor
<b>OSC</b>	Organic Solar Cell
<b>OLED</b>	Organic Light Emitting Diode
<b>ECD</b>	Electrochromic Device
<b>HTL</b>	Hole Transport Layer
<b>ETL</b>	Electron Transport Layer
<b>HIL</b>	Electron Injection Layer
<b>TCO</b>	Transparent Conducting Oxide
<b>PEDOT</b>	Poly (3,4-ethylenedioxythiophene
<b>PSS</b>	Polystyrene sulfonate
<b>BTz</b>	Benzotriazole
<b>ACN</b>	Acetonitrile
<b>TBAF<sub>6</sub></b>	Tetrabutylammoniumhexafluorophosphate

## CHAPTER 1

### 1. INTRODUCTION

#### 1.1 Conjugated Polymers

The conducting ability of polymeric materials are emanating from their  $\pi$  -conjugation between their repeating units that can be clearly defined as extension and delocalization of overlapped  $\pi$ -electrons provide conjugated orientation [1]. Eventhough, semiconducting characteristic of  $\pi$ -conjugated polymers has been known since 1862, the tremendous interest of academia and industry aimed at development of new designed conjugated polymers (CPs) for organic electronics, organic photonics and biosensor has awaken in recent years [2]. The trigger of this growing interest in the field was the discovery of conducting polyacetylene in 1977 by Alan J. Heeger, Alan G. MacDiarmid and Hideki Shirakawa. Later on, the three scientists were awarded by Nobel Prize in Chemistry in 2000 for this excellent discovery. Unlike inorganic semiconductors, conjugated polymers have outstanding advantages like flexibility, low cost, corrosion resistant and processability. Additionally, optical and electronic properties of CPs can be easily altered via synthetic modification [3-5]. Owing to these advantages, CPs have become good candidate to replace inorganic semiconductors. Thanks to their fascinating electronic and optical characteristics and easily tunable morphology, CPs has given rise to enhancement in organic electronics and organic photonics areas such; photovoltaic cells [6], light emitting diodes (LEDs) [7], field effect transistors [8], electrochromics [9] and biosensors [10].



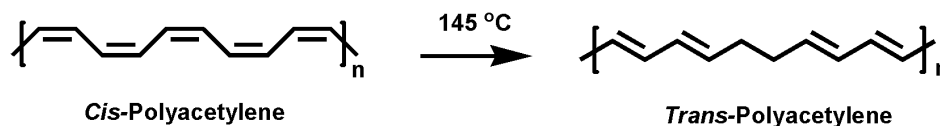
**Figure 1.** Some of the most widely used conjugated polymers

## 1.2 Brief History of Conjugated Polymers

The essential investigations on CPs which gave guidance to today's academic and industrial researches, started in 1960s at two different places in the world. In America, Alan J. Heeger and Alan G. MacDiarmid synthesized an inorganic polymer, sulfur nitride  $(\text{SN})_x$  through solid-state polymerization of  $\text{S}_2\text{N}_2$  crystals. The conductivity measurements at room temperature indicated high results as  $2.5 \times 10^3 \text{ S.cm}^{-1}$ . As the temperature was decreased to 10 K, the conductivity of the polymer was increased over hundreds orders of magnitude. Besides, when both  $\text{Br}_2$  and  $\text{I}_2$  were applied to inorganic polymer,  $10^4$  to  $10^7$  increase in conductivity was observed [11]. This invention opened a new class of materials, conducting polymeric metals [12].



In the meantime, in Japan, polyacetylene (PA) was synthesized directly as a thin film form by mistake by Hideki Shirakawa and his coworkers [13]. The use of thousand times greater Ziegler-Natta catalyst in reaction resulted in insoluble powders. The analysis of infrared (IR) spectra of thin films demonstrated that the stereochemistry of double bonds was affected by the temperature of polymerization. It was observed that at lower temperatures the trans content of PA was diminished. Additionally, at higher temperatures ( $\sim 145\text{ }^{\circ}\text{C}$ ) the cis form converted to trans form irreversibly as shown in **Figure 2**. Electronic studies of cis-trans isomers pointed that the resistivity and energy gap of trans rich PA were  $1.0 \times 10^4\text{ }\Omega\cdot\text{cm}$  and 0.56 eV, respectively. In comparison, cis rich PA had  $2.4 \times 10^8\text{ }\Omega\cdot\text{cm}$  resistivity and 0.93 eV energy gap [13].



**Figure 2.** Irreversible isomerization of polyacetylene

At the end of 1960s Hideki Shirakawa, Alan MacDiarmid and Alan J. Heeger started collaborative studies on behaviour of PA with respect to different oxidants [13,14]. They observed different conductivity for PA through different oxidants. The highest conductivity at room temperature was obtained as  $38\text{ S}\cdot\text{cm}^{-1}$  with treatment with  $\text{I}_2$ .

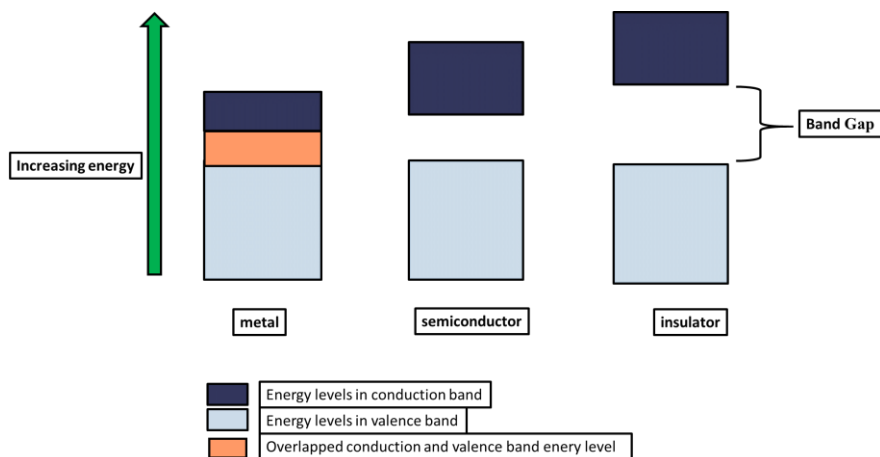
In 1977, the study of oxidatively doped trans-PA, which exhibited metallic conductivity, was reported by these two collaborative research groups [4]. This tremendous discovery is considered as the launch of a new field, plastic electronics

also mentioned as synthetic metals. In 2000, the three scientists were awarded with Nobel Prize in Chemistry “*for the discovery and development of conductive polymers*”.

### 1.3 Band Theory

Solids are classified in three main groups with respect to separation of their occupied and unoccupied energy levels; insulators, semiconductors and conductors. In respect to band theory, the gap between valence band and conduction band of a solid assigns the conductivity. The movement of electrons from valence band (VB) to conduction band (CB) through excitation of electron photochemically or thermally cause conductivity. According to band theory, CB and VB of metals overlapped as a single band hence, the metals have zero band gap [4,15]. The metals, which are also named as conductors, have partially filled bands, which allow movement of free charge carriers resulting in conduction. On the contrary, in insulators, the gap is separated with a large energy thereby the migration of electrons from VB to CB is impossible to conduct electricity. Semiconductors take place in the group between insulators and conductors in terms of conductivity. [16].

The band structure of semiconductors is composed of partially filled valence and empty conduction bands. CPs which have extended  $\pi$ -conjugation between their repeating units, take place in the group of semiconductors. As a result of their extended  $\pi$ -conjugation, CPs can gain ability of conductivity via doping which is described in following sections. As a matter of fact, the  $\pi$ -band of a conjugated polymer is regarded as VB and  $\pi^*$  band is also known as CB.



**Figure 3.** Band gaps of metal, semiconductor and insulator in comparison

#### 1.4 Band gap engineering

Optical and electrochemical properties of the conjugated polymers strongly depend on band gap of the polymers. Therefore, several methodologies have been developed to control band gap and synthesize conjugated polymers with desired properties for certain applications. Bond length alternation (BLA), inductive and mesomeric effect of substituents, chemical rigidification and donor-acceptor approach are the developed strategies for this purpose [17].

Conjugated polymers have two possible resonance structures in their ground state, namely quinoid and aromatic forms. Quinoid form of the polymer is less energetic due to destruction of the aromaticity to adopt this form. Bond length alternation is defined as the ratio of aromatic to quinoid population in a conjugated system and depending on this ratio band gap can be lowered. Conjugated polymers with less aromatic units in the conjugated main chain can easily adopt the quinoid form through  $\pi$ -electron delocalization, which lowers the band gap [17].

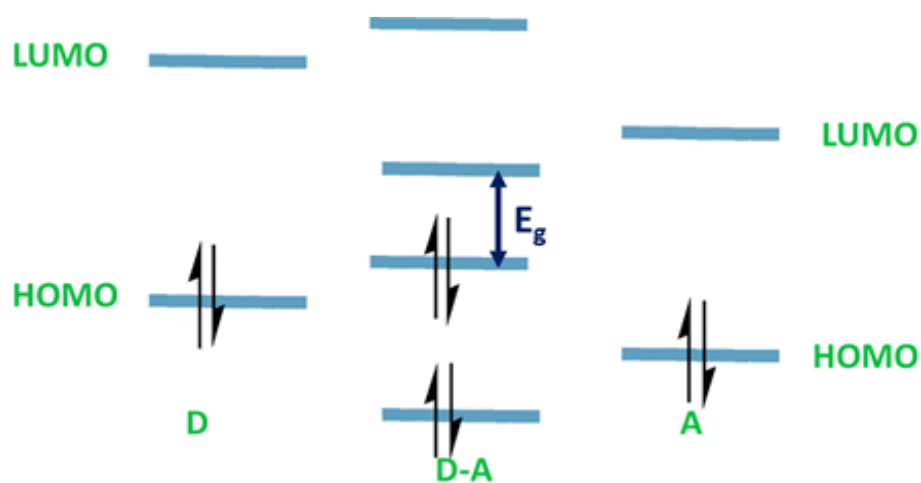
Anchoring electron-donating or electron-withdrawing groups onto the aromatic unit in the main chain is another effective way to control band gap via inductive or mesomeric effects. In general, if electron-donating groups were utilized as substituent, HOMO energy level is raised, while anchoring electron-withdrawing groups lower the LUMO energy. Thus strategy lowers the band gap [17].

Controlling steric and electronic effects on main chain via molecular modification is also useful strategy for band gap engineering. Chemical rigidification is one way to decrease rotational disorder, which increases planarity between adjacent and delocalization. Thus causes decrease in BLA and band gap. Donor- acceptor theory is also another and most effective technique and explained in following section in detail [17].

### **1.5 Donor-acceptor approach**

Alternation of electron rich (donor) and electron deficient (acceptor) units along the same backbone to control electron delocalization, quinoid structures formation and bond length alternation of the conjugated polymers, is known as donor-acceptor approach. This theory was first discovered by Havinga et al; and they showed that different combination of donor and acceptor units strongly affect the band gap of the polymer and a low band gap polymer was synthesized by proper combination of these units [18,19]. Syntheses of lower band gap conjugated polymers compared to parent components are explained by perturbation theory that explain the formation of new HOMO and LUMO energy levels from the interaction of donor and acceptor segments. Interaction of HOMO energy level of donor and HOMO energy level of acceptor cause formation of a new HOMO energy level, which is higher than that of parent components. Similarly, formation of the new LUMO energy level from this interaction

is lower. This causes a decrease in the band gap of polymer compared to their counterparts. Therefore, careful design and selection of the donor and acceptor moieties can provide control over optical and electronic properties. Besides, combination of different donor and acceptor units offer polymers with different band gaps. Therefore, syntheses of novel polymers require a systematic analysis on nature of donor and acceptor units [17].



**Figure 4.** Effect of orbital couplings of donor-acceptor units on band gap

## **1.6. From $\pi$ -Conjugated Polymers to Conducting Polymers: Doping Process and Doping Types**

### **1.6.1 Doping Process**

According to several parameters related to conductivity such as mobility and population density of charge carrier and charge on the electron, CPs are regarded as insulators. However, owing to their semiconductor band structure that allows electronic excitation or electron addition/elimination CPs can gain the ability of conducting. When a charge carrier is introduced to polymer backbone via oxidation or reduction, structure of the chain is distorted in a way that low energy states defined as polaron, bipolaron are formed within bandgap [2]. On account of doping, holes or electrons can move along the polymer backbone to satisfy electrical conductance. By virtue of doping process, conductivity of CPs is increased several orders of magnitude [20]. Both doping and dedoping process are irreversible processes, which do not alter the chemical structure of backbone of CPs.

### **1.6.2 Doping Types**

The mobile charge carriers, which are holes, or electrons that move along the backbone of CPs leading to electrical conductivity, generally do not exist in the backbone of CPs intrinsically. In order to give conducting ability to CPs, these mobile charge carriers can be obtained through two main doping processes:

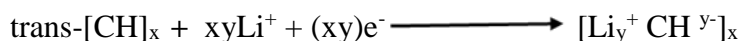
Anionic doping is also named as ‘*p-type doping*’ can be carried out through electrochemical or chemical oxidation. When a positive potential is applied to CP, a dopant anion moves in from the solution into the CP in the direction of the delocalized charge sites on the CP, in this way anionic doping electrochemically arises [21]:



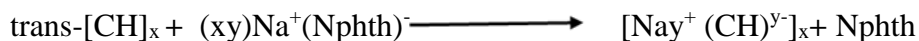
To accomplish anionic doping chemically, CP is exposed to a solution or vapor of an oxidative dopant [22,23]:



Cathodic electrochemical or chemical reduction. Likewise, when a negative potential is applied in solution to CP immobilized on inert electrode, a dopant cation moves in from the solution into the CP in the direction of the delocalized charge sites on the CP, consequently cathodic doping electrochemically occurs [24] :



In order to execute cathodic doping chemically, CP is exposed to a solution or vapor of a reductive dopant [21,23]:



## 1.7 Synopsis of Polymerization Methods

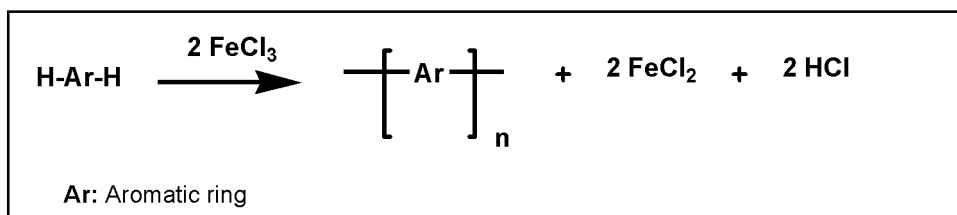
Several polymerization methods can be utilized to synthesize conducting polymers. Chemical and electrochemical polymerization techniques are the most common ones

for polymerization. The chemical polymerization is generally based on Palladium (Pd) catalyzed cross coupling reactions or oxidative ferric chloride polymerization. Electrochemical polymerization involves oxidation of monomers into polymers on electrode surface via applied potential. When two techniques are compared electrochemical polymerization has an advance in time sequence since polymerization occurs in seconds and also the produced polymer is oxidatively doped. On the contrary, chemical polymerization (except oxidative ferric chloride polymerization) requires several hours and synthesized polymer is needed to be doped chemically or electrochemically to their conducting form [25].

## 1.8 Chemical Polymerization

### 1.8.1 Oxidative Ferric Chloride Polymerization

In oxidative ferric chloride polymerization, monomer is polymerized through  $\text{FeCl}_3$  as a reagent as shown in **Figure 5**.



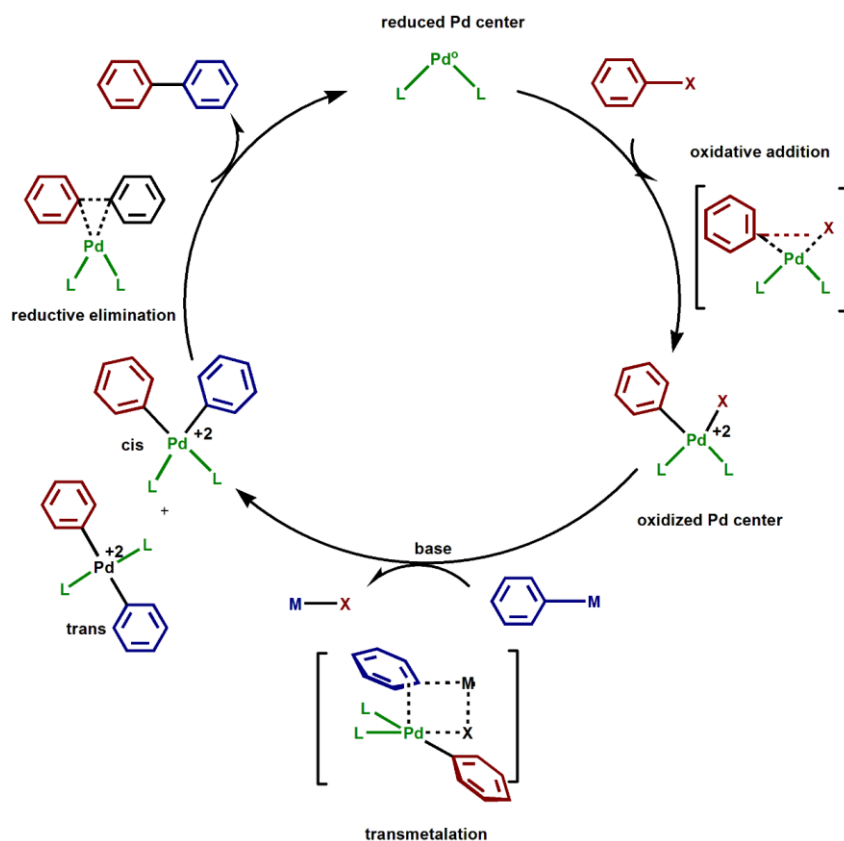
**Figure 5.** Oxidative ferric chloride polymerization



Oxidative polymerization takes place where the hydrogen atom is attached. As a result, it requires no active groups such as stannyl or halide. Oxidative ferric polymerization is preferable in order to attain polymers with high molecular weights [26].

### 1.8.2 Palladium-Catalyzed Cross Coupling Polymerization

Palladium-catalyzed carbon-carbon bond forming reactions have been widely used in organic synthesis due to their relatively mild conditions and tolerance to variety of functional groups. Palladium-catalyzed cross coupling reactions as shown in **Scheme 1** occurs in four main steps; ligand dissociation, oxidative addition of the metal, transmetalation, reductive elimination. In the first step, the catalyst  $ML_n$ , such as  $Pd(PPh_3)_4$ , where M stands for metal ( $Ni^0$ ,  $Pd^0$ ,  $Pt^0$ ) and L represents ligand, dissociates to create the more reactive catalyst with two phosphine groups. Hence,  $Pd(PPh_3)_2$  become the active catalyst. The phosphine groups donate their lone pair to empty d orbital on Pd, resulting  $\sigma$  bonding within phosphine groups and Pd. In this manner, the electron density on Pd is increased hence its nucleophilicity is enhanced. As a result, the second step of the cross coupling reactions, which is oxidative addition becomes inevitable. In this step, an organo-halide (RX) is added to Pd(II) complex through oxidative addition. Transmetalation step involves transfer of nucleophile bonded to organometallic compound from metal to the Pd (II) complex. In the final step, the product is eliminated from Pd catalyst reductively, and the two ligands are transferred to Pd catalyst, therefore Pd catalyst is regenerated  $Pd(0)$  for the next cycle [27-28].

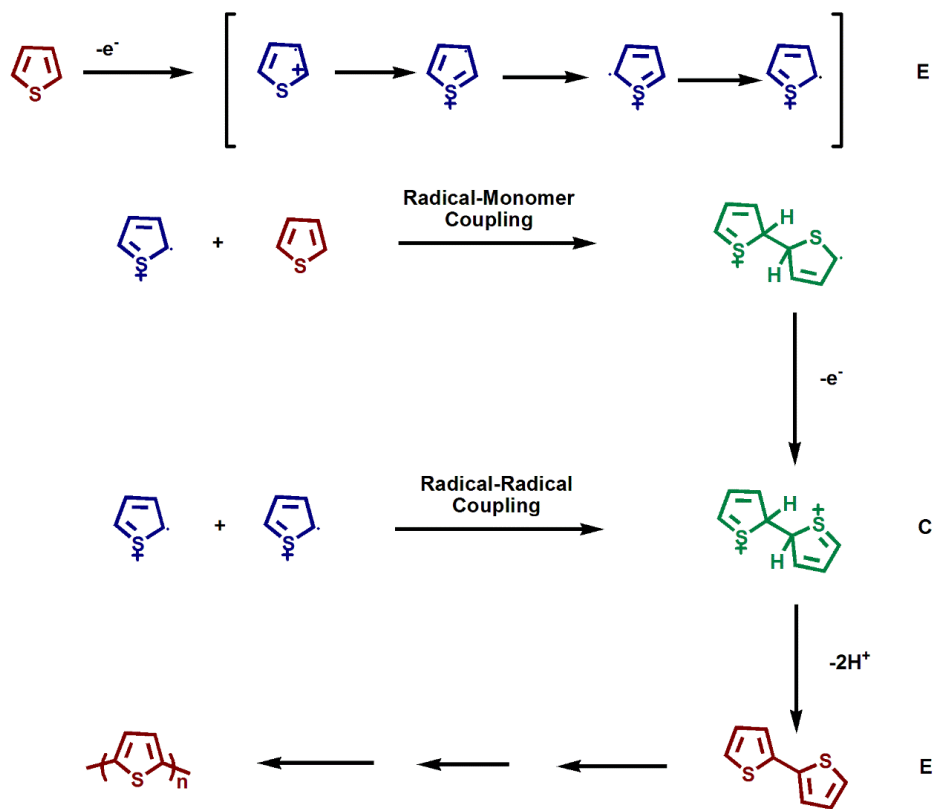


**Scheme 1.** Mechanism of Palladium- catalyzed cross coupling reaction

## 1.9 Electrochemical Polymerization

Electrochemical polymerization involves oxidation of monomers on the surface of indium tin oxide (ITO) coated glass upon applied potential. Electrochemical polymerization provides advantages for further investigations of optical and electrochemical properties of the resulted polymer since it has been already coated on electrode. However, the resulted polymers are in small amounts and also they are generally insoluble. Hence, thermal caharcterizations and molecular weight analysis of the polymers are consequently are imposibble.

During electropolymerization, radical cations are formed and they participate in radical coupling reactions and also electrochemical reactions resulting formation of long polymer chain. The following radical coupling and electrochemical reactions are represented as E(CE)<sub>n</sub> mechanism as shown in **Scheme 2**, representing electrochemical polymerization of thiophene unit [29].



**Scheme 2.** Proposed mechanism of electropolymerization of thiophene

In the electrochemical step of E(CE)<sub>n</sub> mechanism, the monomer is oxidized and radical cation is formed. Electron transfer reaction occurs much faster than monomer diffusion

from bulk solution hence concentration of radical cations is high near the electrode surface. The second step, chemical reaction step includes the coupling reactions between radical cations or radical cation and monomer to form dimer unit in neutral form. Upon applied potential, the dimer unit undergoes oxidation and forms its radical cation via coupling reactions which results in a trimer unit. As polymerization progresses through following electrochemical and chemical reactions, solubility decreases dramatically and the desired polymer precipitates on the surface of electrode [30,31].

### **1.10 Electrochromism**

Chromism is defined as reversible color change upon applied a chemical or physical external stimulus. Chromism is classified according to the applied stimulus resulting in color change such as thermochromism, photochromism, halochromism, solvatochromism, piezochromism and electrochromism. Electrochromism is the reversible and visible color change associated with reduction or oxidation processes caused by applied potential. Different colors are observed due to the generation of different electronic absorption bands during redox processes. The color change is commonly observed between a colored state and a transparent state, or between two colored states. In some cases, formation of several color states can take place due to formation of two redox states, which is named as multichromism [32]. Transition metal oxide, molecular dyes and conducting polymers are three classes of electrochromic materials.

## 1.11 Types of Electrochromic Materials

### 1.11.1 Transition metal oxide films

These kinds of electrochromic materials are classified under inorganic electrochromic materials. Many transition metal oxides in the thin film form like iridium, rhodium, ruthenium, tungsten, manganese, and cobalt exhibit electrochromic property. Tungsten trioxide ( $\text{WO}_3$ ) electrochromic properties were discovered in 1969. After that, it is studied widely.  $\text{WO}_3$  is cathodically coloring material and it is transparent in its neutral state. Blue color forms when it is reduced. The accepted mechanism is the injection and extraction of electrons and metal cations like  $\text{Li}^+$ ,  $\text{H}^+$ , play a crucial role on electrochromic properties of  $\text{WO}_3$  [33].

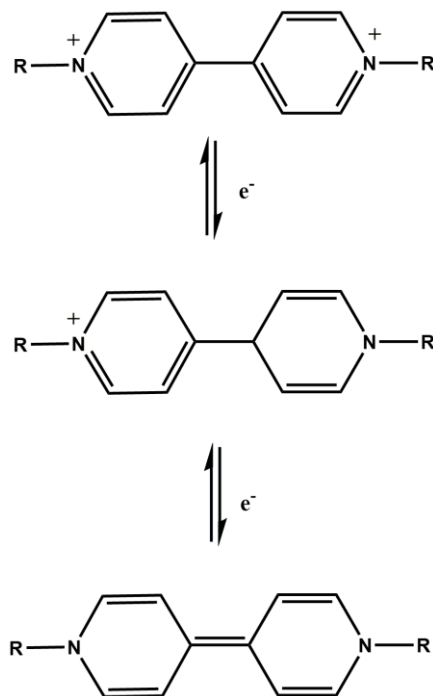


(transparent)      (blue)

### 1.11.2 Viologens

Chemical structure of viologens (1,1'-disubstituted- 4,4'-bipyridilium salts) and their redox processes are depicted in **Scheme 3**. Viologens might have two-step reduction and optical and electrochemical properties of them can be tailored by different conditions like substitution and ion selection. Different alkyl chain or substituents can be anchored to viologens and color of this material is strongly affected by

these changes. For example, if the substituent is *n*-heptyl, a purple color is observed whereas if *p*-cyano-phenyl was the substituent, a green color is observed [32].



**Scheme 3.** Redox process of viologens

### 1.12 Electrochromism in Conducting Polymers

Electrochromic conjugated materials have aroused great attention in the recent years due to their advantages over inorganic counterparts. Electrochromic conjugated polymers have rapid response times, and high optical contrasts and they are more processable compared to inorganic counterparts. Furthermore, color tunability can be done by structure modifications via monomer functionalization and copolymerization.

Electrochromic properties of conjugated polymers are affected by band gap which is the energy difference between valence band and conduction band. Conjugated polymers are insulators in their neutral states. Optical properties of conjugated polymers change by electrochemical or chemical doping and dedoping process. This process causes the introduction of new states (intraband transitions) in the band gap, which is responsible from optical changes and conductivity. Oxidation of conjugated polymers results in the formation of charge carriers like polarons and bipolarons. Therefore, doping process is the main reason for electrochromism in conjugated polymers. Every color of the spectrum can be attained with conjugated polymers [34]. For example, P3HT, polypyrrole and polyaniline switch between red and blue, yellow and black, yellow, green, blue, and violet respectively [35]. The color of neutral state and doped state of conjugated polymers strongly changed by band gap. For example, polymers with band gap greater than 3 eV are transparent in the neutral form and they absorb light in visible region after doping process. Whereas, polymers with band gap equal to or less than 1.5 eV absorb light in visible region and become transparent after doping process. Therefore, factors affecting band gap is the key point to synthesize novel polymers with different neutral state colors

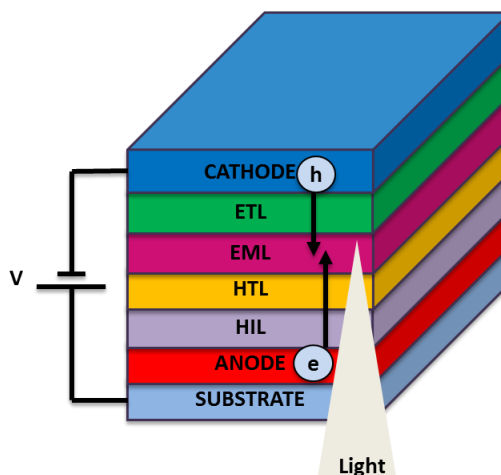
### **1.13. Applications of Conducting Polymers**

The discovery of conducting polymers (CP) has opened a new era where the CPs are regarded as ‘*plastic electronics*’. They have several advantages such as flexibility, processability, low cost. Additionally, they exhibit outstanding electronic and optic properties. They have been used in fabrication of electronic device, solar energy conversion [36], supercapacitors [37], electrochromic devices [38], transistors [39], light emitting diodes [40] and biosensors [41].

### **1.14 Organic Light Emitting Diodes**

Organic light emitting diodes (OLEDs) are solid state light sources consisting of thin film which is an electroluminescent organic material sandwiched between an anode and a cathode. Upon applied potential, electrons and holes are introduced into the electroluminescent material. With the recombination of electrons and holes, excitons are formed which decay by fluorescence with a frequency given by the energy difference between HOMO and LUMO of the electroluminescent material. So overall, light is emitted upon applied potential. Additional thin film layer can be used to utilize hole and electron transport which are named as hole transport layer (HTL) and electron transport layer (ETL). Additionally to enhance the hole transport hole injection layer (HIL) can be introduced to device [42].



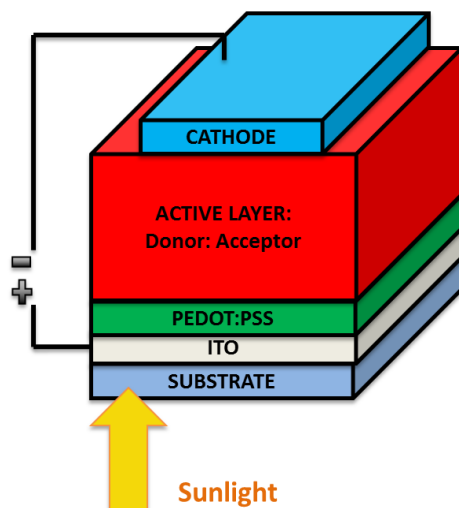


**Figure 6.** Organic light emitting diode device structure

### 1.15 Organic Solar Cells

Organic solar cells (OSCs) are photovoltaic devices that convert the light energy directly to the electrical energy. The main goal to use organic solar cells is the harvesting light energy from the sun. OSCs perform under illumination and produce electrical current that can be directly used or stored chemically or mechanically. OSCs are mainly composed of transparent conducting oxide (TCO) as anode, hole transport layer, active layer (donor: acceptor blend) and cathode. There are five main steps for the operation of organic solar cells: 1) absorption of light by donor material in the active layer, 2) formation of electron and hole pairs which are named as “excitons” 3) diffusion of exciton to the donor-acceptor interface 4) dissociation of exciton, 5) charge collection by the corresponding electrodes. In OSC device configuration, anode is indium tin oxide (ITO), the hole transport layer is poly (3,4-

ethylenedioxythiophene):polystyrene sulfonate (PEDOT:PSS) and the acceptor material is generally a fullerene derivative [43].



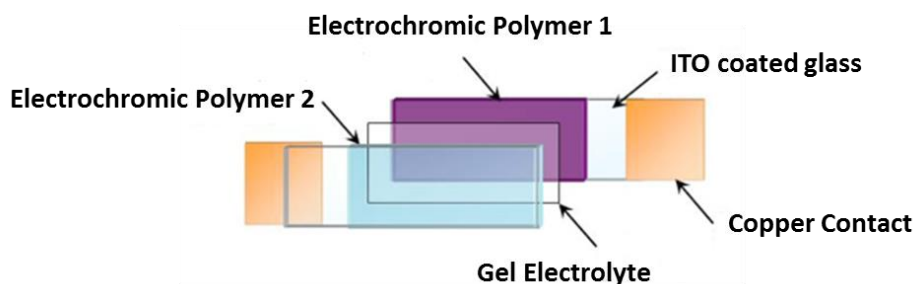
**Figure 7.** Schematic representative of organic solar cell device configuration

### 1.16 Electrochromic Devices

As it is mentioned in detail in the text electrochromism is the reversible and visible color change of the material upon applied potential. Electrochromic conjugated polymers owe good UV-vis absorption, fast switching times, high coloration efficiency and excellent optical contrast besides their color can be tuned through modification of different molecules and groups.

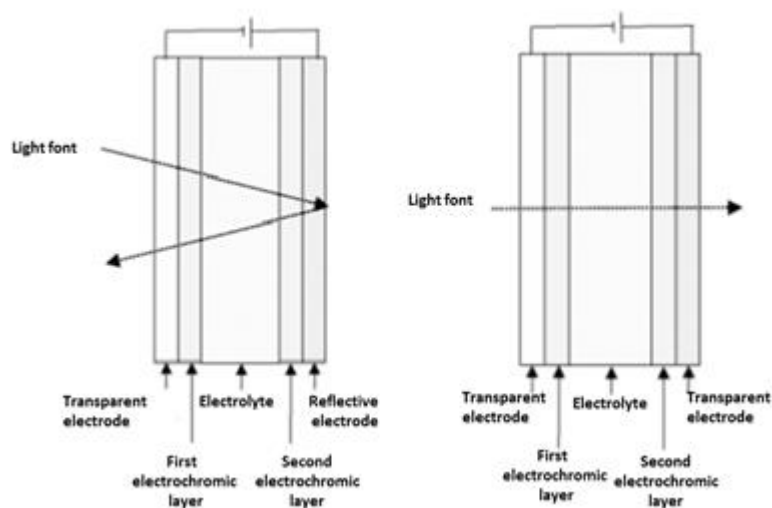
The absorptive/transmissive electrochromic device works through reversible color change of electrochromic material between colored and bleached states. In this device

configuration, both counter and working electrodes are transparent in order to allow to light to pass through the electrochromic device.



**Figure 8.** Schematic absorptive/ transmissive ECD

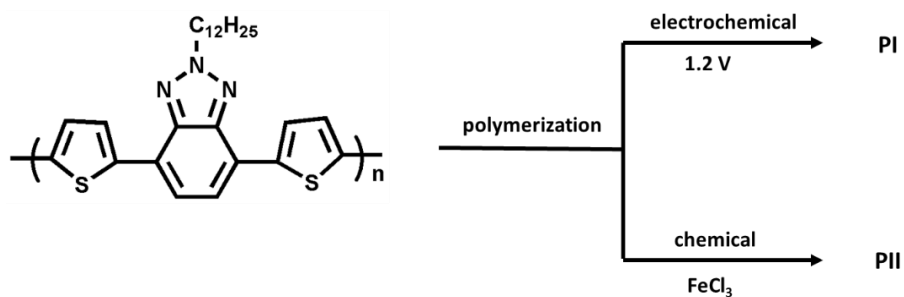
Electrochromic devices allow light and heat control through changing their light transmission properties upon applied potential. Electrochromic devices can work either reflective or transmissive mode. The reflective mode operates to decrease the reflected light such as car rear view mirrors. With this aim, one of the electrical contact is covered with a reflective material such as mirror. In transmissive mode, all layers are transparent, for this reason both electrical contacts are chosen as transparent [44].



**Figure 9.** Schematic representative electrochromic device operating at a) reflective mode and, b) transmissive

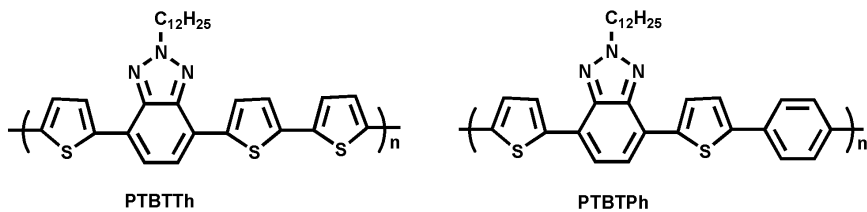
### 1.17 Aim of This Work

With the discovery of conducting polymers both academia and industry have aimed to design and synthesize new CPs for organic electronics, photonics and biosensors. The goal of the researchers in the field of electrochromism is to synthesize polymers which switch between red, blue, green (RGB) colors, black and transmissive. To our research group successfully synthesized a polymer named as **PTBT** that possesses all RGB colors, black and transmissive. Despite its fascinating optical properties; multichromism, good optical contrast, p and n dopability and high stability during kinetic studies, it exhibits relatively high switching time (2 s) [45].



**Figure 10.** Synthesis of **PTBT**

In 2011, two novel benzotriazole (BTz) derivative polymers were synthesized by the same research group to enhance the optical properties and electrochromic properties of **PTBT** (**Figure 11**). The synthesized benzotriazole based polymers exhibited faster switching times than **PTBT**; for **PTBTTh** 1.5 s and for **PTBTPh** 1.6 s.[46].



**Figure 11.** Chemical structures of **PTBTTh** and **PTBTPh**

Up to now, thiophene unit has been commonly utilized as a  $\pi$ -bridge in BTz containing conjugated polymers architecture [47-58]. However, recent studies showed that selenophene substitution in the polymer backbone instead of thiophene is an effective approach to synthesize low band gap polymers with enhanced switching time and conductivity. Selenophene-bearing polymers showed enhanced planarity and extended conjugation length due to its less aromaticity. It was also found that the band gap of selenophene-based polymers was lower compared to other chalcogenophene derivatives because LUMO was predominantly lowered while the HOMO was not affected. These findings were supported by theoretical and experimental findings and attributed to small electronegativity difference between sulphur and selenium and lower ionization potential of selenium [59-66]. Therefore, it is noteworthy to state that usage of selenophene as  $\pi$ -bridge in BTz bearing conjugated polymers can be a useful strategy to synthesize low band gap polymers which also exhibit fast switching time. This study covers the syntheses of two novel polymers; poly(2-(2-octyldodecyl)-4-(selenophen-2-yl)-7-(5-(thiophen-2-yl)selenophen-2-yl)-2H-benzo[d][1,2,3]triazole) (**P-SBTTh**) and poly(2-(2-octyldodecyl)-4-(5-phenylselenophen-2-yl)-7-(selenophen-2-yl)-2H-benzo[d][1,2,3]triazole) (**P-SBTTPh**) containing selenophene as a  $\pi$ -bridge in polymer backbone and also provides a synopsis of basic information about optical, electrochemical and electrochromic properties of polymers.

## CHAPTER 2

### 2. EXPERIMENTAL

#### 2.1 Materials and Methods

Selenophene, n-BuLi, tributyltinchloride, bis(triphenylphosphine)palladium(II) dichloride, n-bromosuccinimide (NBS, benzothiadiazole,), were purchased from Sigma-Aldrich Chemical Co. Ltd.. The commodity chemicals were used as received. Tetrahydrofuran (THF) and toluene were dried over Na/ benzophenone ketyl and freshly distilled prior to use.

#### 2.2 Equipment

$^1\text{H}$  and  $^{13}\text{C}$  NMR spectra were recorded on a Bruker Spectrospin Avance DPX-400 Spectrometer with trimethylsilane (TMS) as the internal reference. The chemical shifts were reported in ppm relative to  $\text{CDCl}_3$  at 7.26 and 77 ppm for the  $^1\text{H}$  and  $^{13}\text{C}$  NMR, respectively. The UV-vis spectra were recorded on Varian Cary 5000 UV-Vis spectrophotometer at room temperature. Cyclic voltammetry studies were carried out in a solution of 0.1 M of tetrabutylammonium hexafluorophosphate in anhydrous

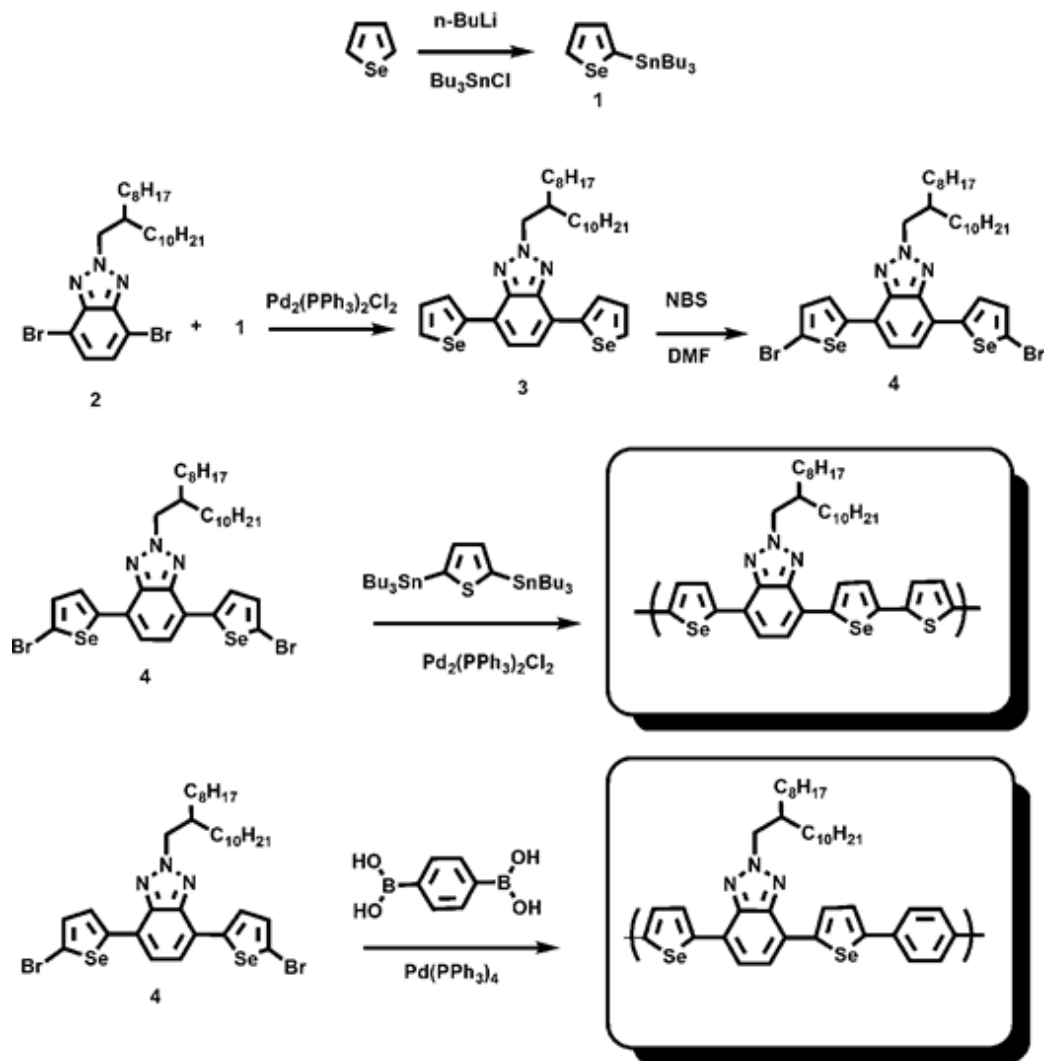
acetonitrile (ACN) at a scan rate of 100 mV/s using a Voltalab 50 potentiostat in a three-electrode cell consisting of an ITO-coated glass slide as the working electrode, Pt wire as the counter electrode, and Ag wire as the pseudo reference electrode (calibrated against Fc/Fc<sup>+</sup> (0.3 V)). Thermal properties of the polymers were analyzed by thermogravimetry analysis (TGA) (Perkin Elmer Pyris 1 TGA) under N<sub>2</sub> atmosphere with a heating rate of 10 °C/min up to 300°C. Average molecular weight of the polymer was determined by gel permeation chromatography (GPC) using a Polymer Laboratories GPC 220.

## 2.3 Synthesis

Synthetic route for the monomers and polymers were depicted in **Scheme 4**. The same procedure was applied for the synthesis of 4,7-dibromo-2-(2-octyldodecyl)-2*H*-benzo[1,2,3]triazole (**2**), which was described in detail in our previously published study [58]. BTz was anchored with long and branched solubilizing alkyl chain (2-octyldodecane) as the pendant group to ensure the solubility of the resulting polymers. Compound **2** was then coupled with tributyl(selenophen-2-yl)stannane (**1**) via Stille coupling to afford 2-(2-octyldodecyl)-4,7-di(selenophen-2-yl)-2*H*-benzo[d][1,2,3]triazole (**3**). Bromination of **3** was performed in the presence of NBS and DMF to synthesize 4,7-bis(5-bromoselenophen-2-yl)-2-(2-octyldodecyl)-2*H*-benzo[d][1,2,3]triazole (**4**). Then **4** was utilized for the syntheses of polymers. **P-SBTTh** was synthesized by Stille coupling copolymerization of **4** with 2,5-bis(tributylstannyl)thiophene in the presence of Pd<sub>2</sub>(PPh<sub>3</sub>)<sub>2</sub>Cl<sub>2</sub> in THF solvent. **P-SBTPh** was synthesized by Suzuki coupling using **4** and 1,4-phenylenediboronic acid in the presence of Pd(PPh<sub>3</sub>)<sub>4</sub> in toluene-water solvent system. The monomer and

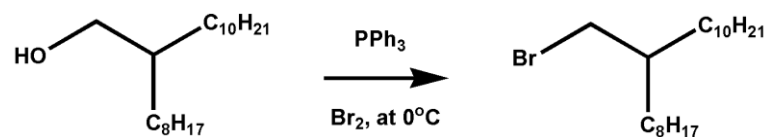


copolymer molecular structures were confirmed by NMR spectroscopy. After successive purification via Soxhlett extraction with methanol, acetone and chloroform.



**Scheme 4.** Synthetic pathways of monomer and polymers

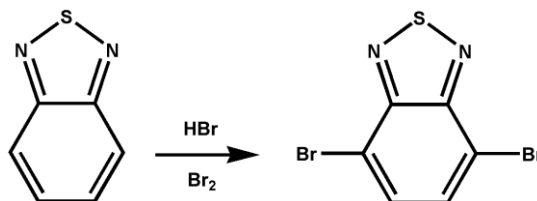
### 2.3.1 Synthesis of 9-(bromomethyl)nonadecane



**Figure 12.** Synthesis of 9-(bromomethyl)nonadecane

2-Octyl-1-dodecanol (1.94 g, 6.50 mmol) was dissolved in  $\text{CH}_2\text{Cl}_2$  (DCM) (50 mL) and triphenylphosphine ( $\text{PPh}_3$ ) (1.79 g, 6.82 mmol) was added to solution at  $0^\circ\text{C}$ . Bromine (0.35 mL, 15.5 mmol) was added to reaction mixture and reaction stirred for half an hour at  $0^\circ\text{C}$ . Then, reaction was warmed to room temperature and reaction mixture was washed with  $\text{NaHSO}_3$  and organic layer was further washed with distilled water and brine. The organic layer was dried over  $\text{MgSO}_4$  and the solvent was evaporated under reduced pressure. Column chromatography on silica gel using hexane yielded colorless oil (2.23 g, 94%).  $^1\text{H}$  NMR (400 MHz,  $\text{CDCl}_3$ ), d (ppm): 3.37(d,  $J= 4.7$  Hz, 2H), 1.52 (m, 1.46–1.57, 1H), 1.20 (m, 1.15–1.32, 32 H), 0.81(t,  $J =6.58$  Hz, 6H).  $^{13}\text{C}$  NMR (400 MHz,  $\text{CDCl}_3$ ), d (ppm): 39.6, 39.5, 32.59, 31.9, 29.8, 29.7, 29.6, 29.5, 29.4, 29.3, 26.6, 22.7, 14.1.

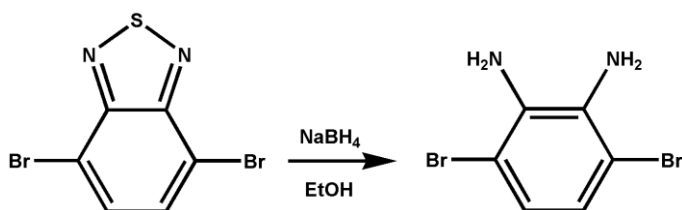
### 2.3.2 Synthesis of 4,7-dibromobenzo[c][1,2,5]thiadiazole



**Figure 13.** Synthesis of 4,7-dibromobenzo[c][1,2,5]thiadiazole

Benzo-1,2,5-thiadiazole (5.00 g, 36.7 mmol) was dissolved in HBr (47%)(30 mL). After, a solution of Br<sub>2</sub> (17.6 g, 110 mmol) in HBr (30 mL) was added slowly to reaction mixture at room temperature. After addition was completed, the reaction mixture was refluxed at 135 °C for overnight. Then the mixture was cooled to room temperature and filtered to obtain an orange solid residue. The residue was washed with NaHSO<sub>3</sub> solution to consume excess Br<sub>2</sub>. The crude was then dissolved in DCM and washed with brine. The organic layer was dried over MgSO<sub>4</sub> and the solvent was evaporated under reduced pressure. The obtained solid was washed with cold diethyl ether several times to afford 4,7-dibromobenzo[c][1,2,5]thiadiazole (9.7 g, yield 90%) as a yellow solid.

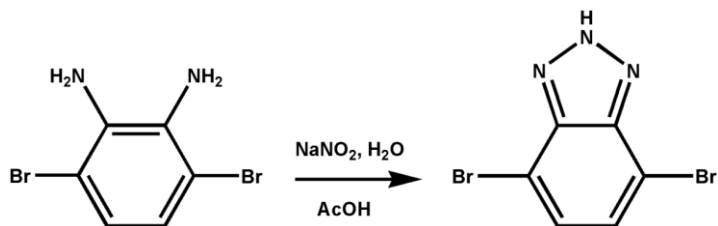
### 2.3.3 Synthesis of 3,6-dibromobenzene-1,2-diamine



**Figure 14.** Synthesis of 3,6-dibromobenzene-1,2-diamine

4,7-Dibromobenzothiadiazole (5.0 g, 17 mmol) was dissolved in ethanol (EtOH) (50 mL) in 1000 mL round bottom flask. The mixture was cooled to 0°C in an ice bath and  $\text{NaBH}_4$  powder (25.7 g, 0.68 mol) was slowly added to reaction mixture. When gas evolution was stopped, the reaction was warmed to room temperature and stirred for 20 hours. The solvent was evaporated and the residue was partitioned between diethyl ether and brine. The organic phase was separated and the solvent was evaporated under reduced pressure to yield a faint yellow solid, 3,6-dibromobenzene-1,2-diamine (3.5 g, yield 80%).

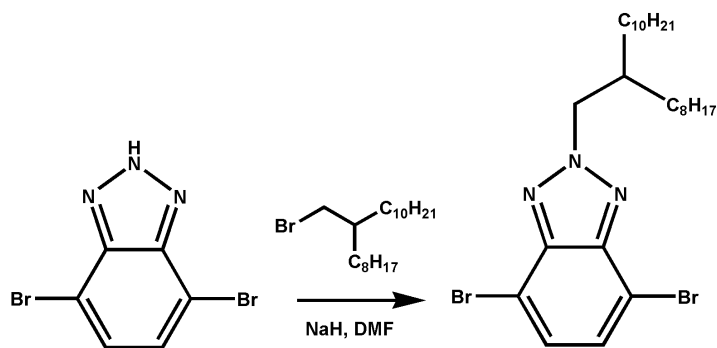
### 2.3.4 Synthesis of 4,7-dibromo-2H-benzo[d][1,2,3]triazole



**Figure 15.** Synthesis of 4,7-dibromo-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole

To a solution of 3,6-dibromobenzene-1,2-diamine (5.00 g, 18.8 mmol) in 75 mL acetic acid (AcOH), a solution of  $\text{NaNO}_2$  (1.88 g, 20.7 mmol) in 36 mL  $\text{H}_2\text{O}$  was added slowly and the mixture was stirred for 20 minutes at room temperature. The precipitate was collected by filtration and washed with distilled water several times to afford 4,7-dibromo-2H-benzo[d][1,2,3]triazole as a pink powder (2.51 g, yield 48%).

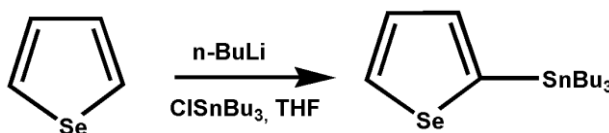
### 2.3.5 Synthesis of 4,7-dibromo-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole



**Figure 16.** Synthesis of 4,7-dibromo-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole

4,7-Dibromo-2H-benzo[d][1,2,3]triazole (2.5 g, 9.0 mmol) was dissolved in dimethylformamide (DMF) (10 mL). Then, NaH (264 mg, 11.0 mmol) was added to reaction mixture at 0°C and 9-(bromomethyl)nonadecane (3.45g, 9.56 mmol) was added at this temperature. The reaction was stirred at room temperature overnight. Water was added and the reaction mixture was extracted with brine. The organic residue was dried over MgSO<sub>4</sub> and then, evaporation was done under reduced pressure. Column chromatography on silica gel with hexane and CHCl<sub>3</sub>(1:2) yielded 4,7-dibromo-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole as a faint yellow oil (2.3 g, yield 40%). <sup>1</sup>H NMR (400 MHz, CDCl<sub>3</sub>), d (ppm): 7.36 (s, 2H), 4.62 (d, J = 7.3 Hz, 2H), 2.26 (m, 1H), 1.15 (m, 32H), 0.80 (m, 6H). <sup>13</sup>C NMR (400 MHz, CDCl<sub>3</sub>), d (ppm): 140.3, 128.2, 106.3, 56.6, 35.3, 29.3, 29.4, 29.3, 29.0, 27.5, 27.3, 27.2, 27.1, 27.0, 26.9, 23.5, 20.4, 20.3, 11.5.

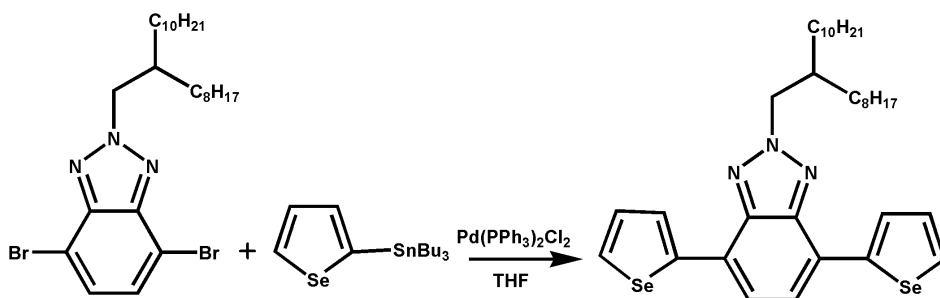
### 2.3.6 Synthesis of Tributyl(selenophen-2-yl)stannane



**Figure 17.** Synthesis of Tributyl(selenophen-2-yl)stannane

Selenophene (1.0, 7.6 mmol) was dissolved in dry tetrahydrofuran (THF) (20 mL) and the reaction mixture was cooled to  $-78^\circ\text{C}$  under argon atmosphere. 2.5 M  $n\text{-BuLi}$  in hexane (3.66 mL, 39.6 mmol) was dropwisely added to reaction mixture. After addition of  $n\text{-BuLi}$  was completed, the reaction mixture was stirred for 1 hour at  $-78^\circ\text{C}$ . Then, stannyltributyltinchloride (2.21 mL, 8.15 mmol) was slowly added to the reaction mixture at  $-78^\circ\text{C}$ . After addition was completed, the reaction mixture was stirred for 1.5 hours at  $-78^\circ\text{C}$ , warmed to room temperature and stirred overnight. The reaction mixture was poured to saturated solution of  $\text{NaHCO}_3$  and organic phase was separated, washed with brine and dried over  $\text{MgSO}_4$ . The residue was purified by neutral alumina column chromatography with hexane as the eluent to obtain colorless oil of tributyl(selenophen-2-yl)stannane (2.56 g, yield 80%).  $^1\text{H}$  NMR (400 MHz,  $\text{CDCl}_3$ ),  $\delta$  (ppm): 8.5 (d,  $J=4.8$ , 1H), 7.7 (m,  $J=7.3$  Hz, 2H), 1.8 (m, 6H), 1.55 (m, 6H), 1.1 (m, 6H), 0.9 (t, 9H)

### 2.3.7 Synthesis of 2-(2-octyldodecyl)-4,7-di(selenophen-2-yl)-2H-benzo[d][1,2,3]triazole

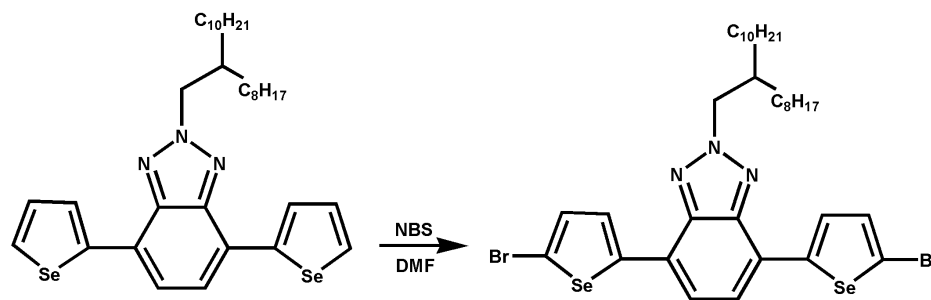


**Figure 18.** Synthesis of 2-(2-octyldodecyl)-4,7-di(selenophen-2-yl)-2H-benzo[d][1,2,3]triazole

4,7-Dibromo-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole (852 mg, 1.53 mmol) and tributyl(selenophen-2-yl)stannane (2.8 g, 6.7 mmol) were dissolved in THF (100 mL). Bis(triphenylphosphine)palladium(II)dichloride (135 mg, 0.19 mmol) was added into the solution and the reaction mixture was refluxed for 18 h under argon atmosphere. After the reaction was completed, solvent was removed under reduced pressure. Column chromatography was performed by silica gel with petroleum ether and DCM (10:1) and product was obtained as a yellow solid (790 mg, yield % 78).  $^1\text{H}$  NMR (400 MHz,  $\text{CDCl}_3$ ),  $\delta$  (ppm): 8.1 (dd,  $J_1=0.95$  Hz,  $J_2=2.90$  Hz, 2H), 7.9 (dd,  $J_1=0.95$  Hz,  $J_2=4.68$ , 2H), 7.5 (s, 2H), 7.32 (dd,  $J_1=1.73$  Hz,  $J_2=3.89$  Hz, 2H), 4.64 (d,  $J=6.42$  Hz, 2H), 2.18 (m, 2.14–2.23, 1H), 1.25 (m, 1.12–1.32, 32H), 0.78 (m, 0.74–0.82, 6H).  $^{13}\text{C}$  NMR (400 MHz,  $\text{CDCl}_3$ ),  $\delta$  (ppm): 143.7, 140.3, 130.0, 128.9, 126.6, 123.9, 121.3, 75.6, 75.3, 37.7, 30.5, 30.4, 30.1, 28.4, 28.2, 28.1, 27.9, 24.9, 21.3, 12.7.



### 2.3.8 Synthesis of 4,7-bis(5-bromoselenophen-2-yl)-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole

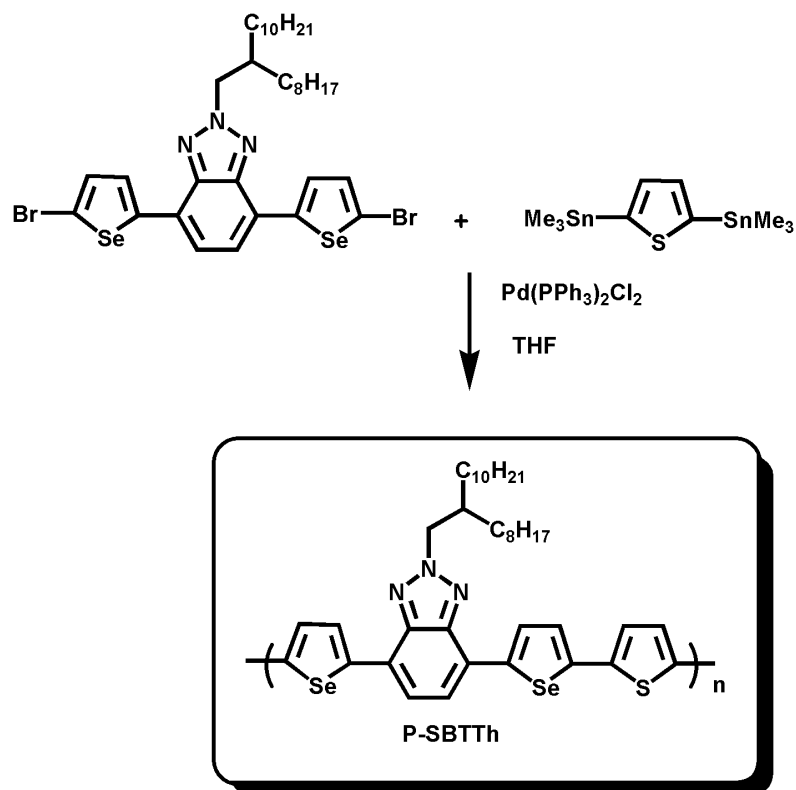


**Figure 19.** Synthesis of 4,7-bis(5-bromoselenophen-2-yl)-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole

2-(2-Octyldodecyl)-4,7-di(selenophen-2-yl)-2H-benzo[d][1,2,3]triazole (790 mg, 1.20 mmol) was dissolved in anhydrous DMF (30 ml). N-bromosuccinimide (517 mg, 3.10 mmol) was added and the mixture was stirred at room temperature under argon atmosphere for 18 hours. The solvent was removed by under reduced pressure, the R-residue was dissolved in  $\text{CHCl}_3$  and washed with water. The organic layer was separated, dried with  $\text{MgSO}_4$  and the solvent was evaporated. Column chromatography was performed by silica gel with  $\text{CHCl}_3$  and the product was obtained as a dark yellow solid (957 mg, yield % 97).  $^1\text{H}$  NMR (400 MHz,  $\text{CDCl}_3$ ),  $\delta$  (ppm): 7.7 (d,  $J=4.22$  Hz, 2H) 7.46 (s, 2H) 7.25 (d,  $J=4.22$  Hz, 2H) 4.65 (d,  $J=6.42$  Hz, 2H), 2.18 (m, 2.14–2.22, 1H), 1.25 (m, 1.12–1.35, 32H), 0.79 (m, 0.73–0.85, 6H).  $^{13}\text{C}$  NMR (400 MHz,  $\text{CDCl}_3$ ),  $\delta$  (ppm): 132.6, 126.3, 133.5, 132.6, 130.7, 129.0, 126.1, 114.8, 77.3, 77.0, 76.7, 31.9, 31.5, 29.9, 29.6, 29.3, 26.3, 22.7, 14.1.

### 2.3.9 Synthesis of P-SBTTh

4,7-Bis(5-bromoselenophen-2-yl)-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole (340 mg, 0.42 mmol) and 2,5-bis(tributylstannyl)thiophene (280 mg, 0.42 mmol) were dissolved in anhydrous THF under argon atmosphere and heated to reflux. Bis(triphenylphosphine)palladium(II)dichloride (0.125 g) was added to reaction mixture and the reaction was refluxed for three days. Polymer was precipitated with methanol and Soxhlett extraction was carried out with acetone and hexane. Polymer was recovered by chloroform and dried under vacuum. **P-SBTTh** was obtained as a purple solid with a yield of 35.4 % (110 mg). GPC: number average molecular weight (Mn): 2500, molecular average molecular weight (Mw): 3000, polydispersity index (PDI): 1.2. <sup>1</sup>H NMR (400 MHz, CDCl<sub>3</sub>), δ (ppm): 8.02 (BTz), 7.64 (Thiophene), 7.47 (Selenophene), 7.37 (Selenophene) 4.7 (N-CH<sub>2</sub>, BTz), 2.24 (-CH), 1.4-0.9 (-CH<sub>2</sub>), 0.76 (-CH<sub>3</sub>)

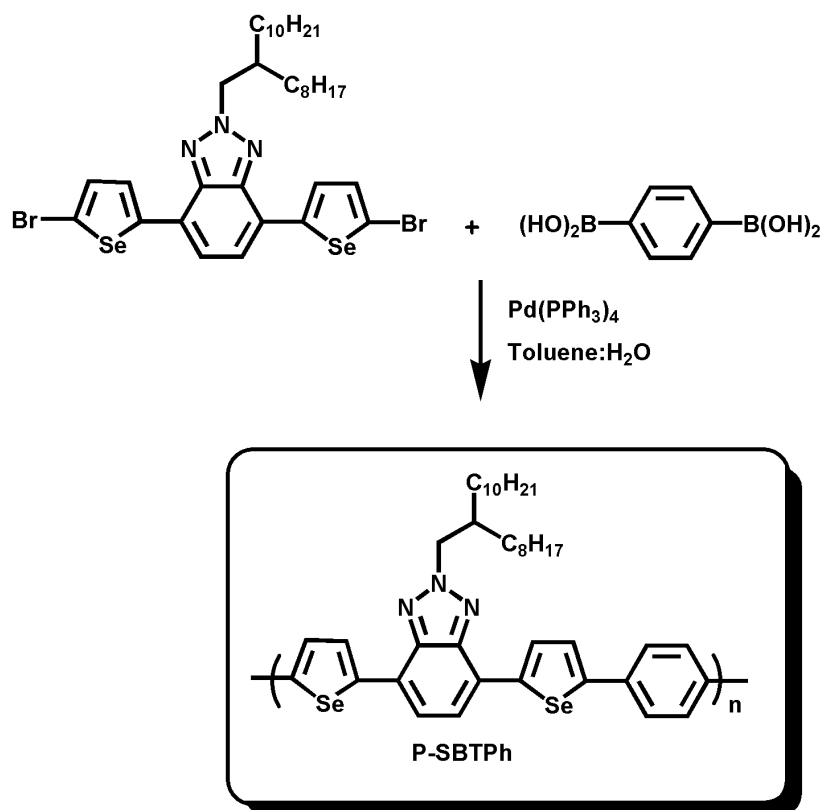


**Figure 20.** Synthesis of **P-SBTTh**

### 2.3.10 Synthesis of P-SBTPh

4,7-Bis(5-bromoselenophen-2-yl)-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole (670 mg, 0.82 mmol) and 1,4-phenylenediboronic acid (136 mg, 0.82 mmol) potassium carbonate ( $\text{K}_2\text{CO}_3$ , 2 M in  $\text{H}_2\text{O}$ ), toluene (3: 2 toluene : water) (50 mL),  $\text{Pd(PPh}_3)_4$  (5 mol %), and tetrabutylammonium iodide ( $\text{N(Bu)}_4\text{I}$ , (1 mol %) were refluxed under argon atmosphere for 24 hours. The solvent was removed and the residue was poured into water and extracted with  $\text{CHCl}_3$ . The solvent was evaporated under reduced pressure. Polymer was precipitated in methanol and Soxhlett extraction was carried out

with acetone and hexane. Polymer was recovered by chloroform and dried under vacuum. **P-SBTPh** was obtained as a purple solid with a yield of 16.6 % (100 mg). GPC: Number Mn: 15400, Mw: 19200, PDI: 1.25.  $^1\text{H}$  NMR (400 MHz,  $\text{CDCl}_3$ ),  $\delta$  (ppm): 8.03 (BTz), 7.64 (Selenophene), 7.57 (Selenophene), 7.3 (Benzene), 4.72 (N-CH<sub>2</sub>), 2.26 (-CH), 1.3-1.0 (-CH<sub>2</sub>), 0.9-0.7 (-CH<sub>3</sub>).



**Figure 21.** Synthesis of **P-SBTPh**

## **2.4 Characterization of Conducting Copolymers**

### **2.4.1 Gel Permeation Chromatography**

Gel permeation chromatography (**GPC**) is a chromatographic technique that aims to determine the molecular weight of the polymers and it involves solid stationary phase and liquid mobile phase. GPC differs from other chromatographic techniques with its separation mechanism. The separation mechanism of GPC relies on the size of the polymer molecules in solution rather than the chemical interactions between stationary phase and particles.

### **2.4.2 Thermal Analysis**

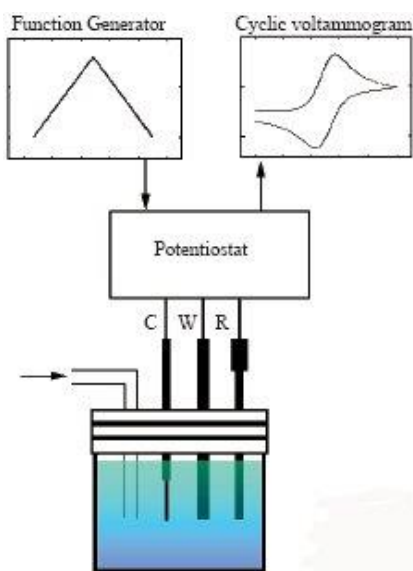
Thermal properties of the polymers were examined by thermal gravimetry analysis (TGA) and differential scanning calorimetry (DSC) under nitrogen atmosphere. DSC studies were performed with Perkin Elmer Differential Scanning Calorimetry. In DSC analysis, heat capacity of polymer by temperature change was examined. DSC analysis allows detection of glass transition temperature ( $T_g$ ), melting temperature ( $T_m$ ).

The mass of polymer decreases upon heating and TGA studies monitor the mass change as a function of temperature. TGA studies give information about decomposition temperature of polymer ( $T_d$ ) and quantity of weight loss. Thermal gravimetric analysis was performed with Perkin Elmer Pyris 1 Thermal Gravimetry Analysis.

### 2.4.3 Cyclic Voltammetry

Cyclic voltammetry (CV) is a multi-purpose electroanalytical technique that examines the electroactivity of compounds and also CV is used for the electropolymerization of monomers. CV gives information about p or n type doping potentials and their corresponding dedoping potentials of electroactive materials.

CV comprises of cycling the potential applied onelectrode immersed in solution and measuring the resulting current. The potential of working electrode is controlled with respect to a reference electrode. The controlled potential utilized across the reference and working electrodes is the initiation for the polymerization. The excitation signal is a linear potential with a triangular waveform as shown in **Figure22**. The potential sweeps between any two values; as switching potentials. The current response is monitored as a function of applied potential, **Figure22** represents a simple cyclic voltammogram.



**Figure 22.** Representative cyclic voltammetry (CV)

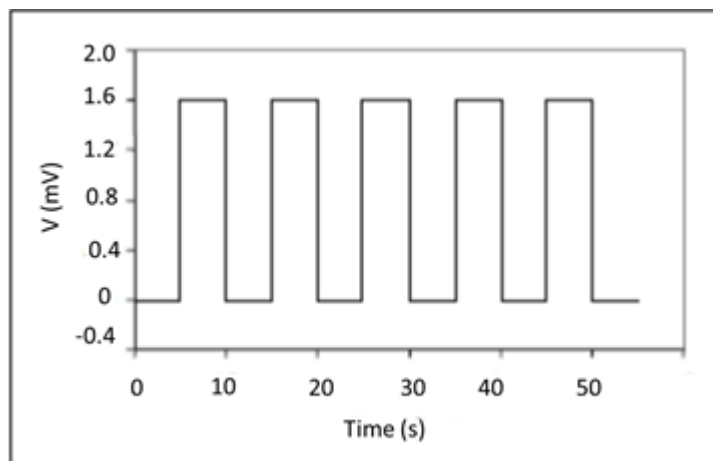
#### 2.4.4 Spectroelectrochemistry

Spectroelectrochemical analysis combines the spectroscopic and electrochemical analyses. In order to demonstrate the color and the color changes that occur upon doping processes, spectroelectrochemistry analysis is utilized. In addition, spectroelectrochemistry analysis gives information on maximum absorption wavelength ( $\lambda_{\text{max}}$ ), optical band gap ( $E_g^{\text{opt}}$ ) and intergap states named as bipolarons and polarons.

To perform spectroelectrochemical analysis, a polymer coated ITO (indium tin oxide) is placed in a cuvette which involves a reference electrode ( $\text{Ag}/\text{Ag}^+$ ) and Pt wire as the counter electrode. The set up cell is then combined with a potentiostat and the polymer is oxidized upon applied potential, simultaneously the absorption spectrum of the polymer is recorded.

#### 2.4.5 Kinetic studies

Switching time and optical contrast are important parameters for an efficient electrochemical device of conducting polymers. To investigate the switching time and optical contrast of the polymer, a square wave potential step coupled with spectroscopy method known as chronoamperometry was applied. The transmittance change between oxidized and reduced states is named as ‘*optical contrast*’. The time is needed for a polymer to oxidize from its reduced state is known as ‘*switching time*’.



**Figure 23.** Representative square wave voltammetry



## CHAPTER 3

### 3. RESULTS AND DISCUSSION

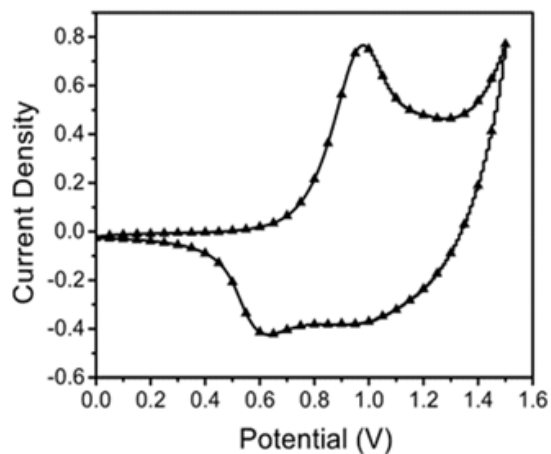
#### 3.1 Electrochemical Characterization of Copolymers

##### 3.1.1 Electrochemical Properties of P-SBTTh

Cyclic voltammetry is a versatile instrument utilized to determine redox behavior and HOMO and LUMO energy levels of the conjugated polymers. Therefore, electronic behavior of **P-SBTTh** was determined by cyclic voltammetry (CV). **P-SBTTh** was dissolved in chloroform (5 mg/mL) and spray coated on ITO electrode. Then CV was performed in three-electrode cell consisting of an ITO-coated glass slide as the working electrode and a Pt wire as the counter electrode. CV measurements were carried out in 0.1 M TBAPF<sub>6</sub> /ACN solution at a scan rate of 100 mV/s (**Fig. 24**). Electrochemical properties of polymer were summarized in **Table 1**.

**P-SBTTh** showed only p-doping property. **P-SBTTh** showed one oxidation potential at 0.93 V and a reversible reduction couple at 0.62 and 0.94 V. HOMO energy level of **P-SBTTh** was determined from onset potential of oxidation, which was observed at 0.69 V. HOMO energy level of **P-SBTTh** was calculated from  $\text{HOMO} = (4.75 + E_{\text{onset}}^{\text{ox}})$  and found as -5.44 eV, respectively. Relative LUMO energy level of polymer was

calculated by using optical band gap and HOMO energy level due to absence of n-doping and determined as -3.9 eV.



**Figure 24.** Cyclic voltammograms of **P-SBTTh** in 0.1 M TBAPF<sub>6</sub>/ACN solvent-electrolyte couple with a scan rate of 100 mV/s

**Table 1.** Summary of optical and electrochemical properties of **P-SBTTh**

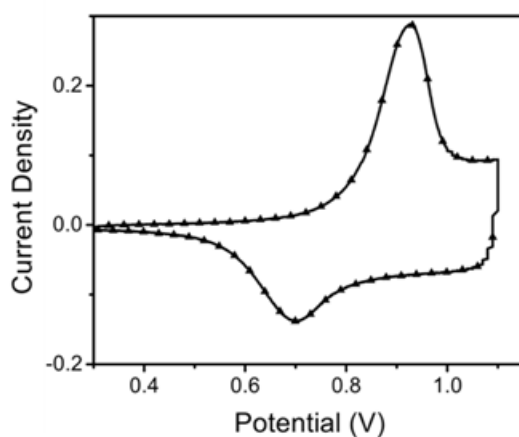
Electrochemical Properties			Optical Properties		
Polymer	HOMO (eV)	LUMO* (eV)	Thin film $\lambda_{\text{max}}$ (nm)	Solution $\lambda_{\text{max}}$ (nm)	$E_{\text{g}}^{\text{opt}}$ (eV)
<b>P-SBTTh</b>	-5.44	-3.90	550	544	1.54

\*LUMO energy level is calculated from optical data.

### 3.1.2 Electrochemical Properties of P-SBTPh

Electronic behavior of **P-SBTPh** was determined by cyclic voltammetry (CV). Polymers were dissolved in chloroform (5 mg/mL) and spray coated on ITO electrode. Then CV was performed in three-electrode cell consisting of an ITO-coated glass slide as the working electrode, Pt wire as the counter electrode. CV measurements were carried out in 0.1 M TBAPF<sub>6</sub> /ACN solution at a scan rate of 100 mV/s (**Fig. 25**). Electrochemical properties of polymer was summarized in **Table 2**.

**P-SBTPh** also showed only p-doping property. Although, **P-SBTTh** has thiophene ring in its backbone that is more electron rich compared to benzene, **P-SBTPh** oxidation potential was also observed at 0.93 V. HOMO energy level of **P-SBTPh** was determined from onset potential of oxidation, which was observed at 0.79 eV. HOMO energy level of **P-SBTPh** was calculated from  $\text{HOMO} = (4.75 + E_{\text{onset}}^{\text{ox}})$  and found as -5.54 eV, respectively. Relative LUMO energy level of polymer was calculated using optical band gap and HOMO energy level due to absence of n-doping and determined -3.92 eV.



**Figure 25.** Cyclic voltammograms of **P-SBTPh** in 0.1 M TBAPF<sub>6</sub>/ACN solvent-electrolyte couple with a scan rate of 100 mV/s.

**Table 2.** Summary of optical and electrochemical properties of **P-SBTPh**

Electrochemical Properties			Optical Properties		
Polymer	HOMO (eV)	LUMO* (eV)	Thin film $\lambda_{\text{max}}$ (nm)	Solution $\lambda_{\text{max}}$ (nm)	$E_{\text{g}}^{\text{opt}}$ (eV)
<b>P-SBTPh</b>	-5.54	-3.92	537	516	1.62

\*LUMO energy level is calculated from optical data.

When we compared oxidation potential of **P-SBTTh** and **P-SBTPh** with their thiophene counterparts, namely **P-TBTTh** and **P-TBTPh**, their oxidation potentials were observed at lower potentials. Oxidation potentials of **P-TBTTh** and **P-TBTPh** were 1.04 and 1.12 eV respectively[55]. Difference in oxidation potentials can be attributed to electron rich nature of selenophene compared to thiophene. Although **P-TBTTh** and **P-TBTPh** bear dodecyl group as the side chain, this does not hinder the comparison since substitution of branched alkyl chain has no effect on optical and electrochemical properties whereas it mostly affects physical properties [68,69]. Therefore, most of the changes in electrochemical properties are the result of different chalcogenophene rings in polymer structure.

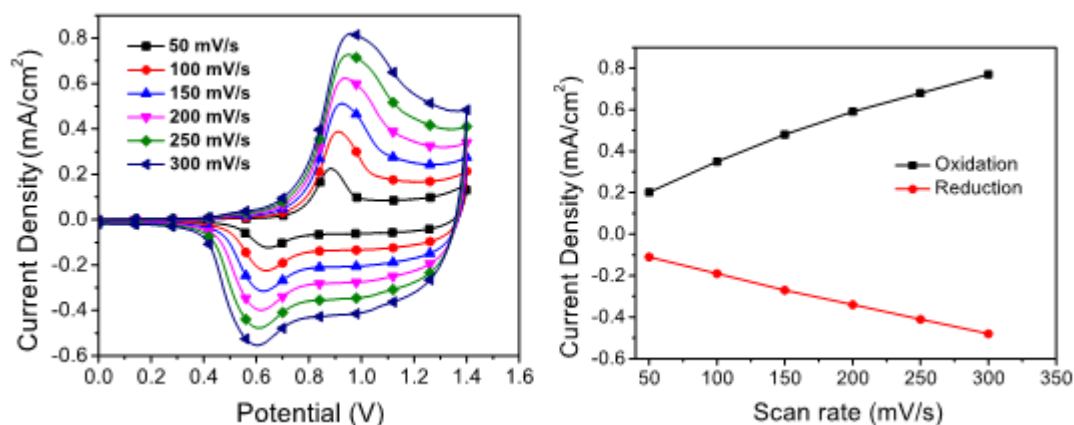
### 3.1.3 Scan Rate Studies of Polymers

The scan rate dependence of the anodic and cathodic peak currents was studied in a electrolyte solution at different scan rates (**Figure 26** and **Figure 27**). In this system, diffusion from solution to the surface area of the polymer does not take place and it is

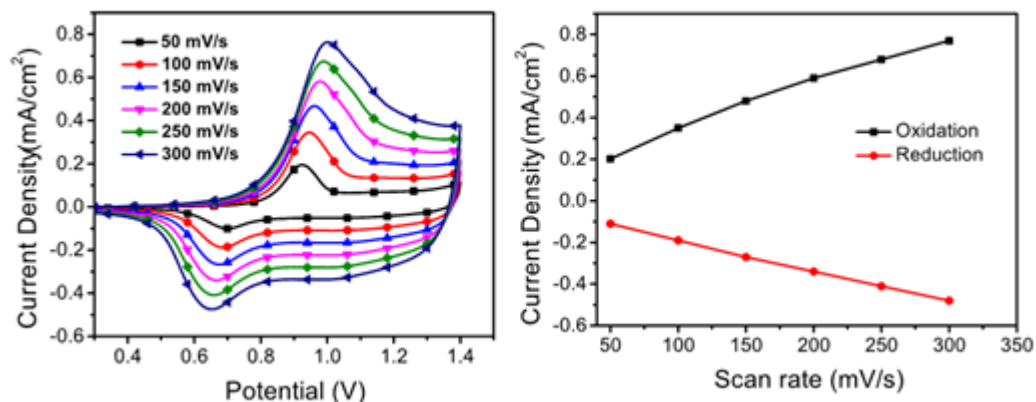
constant. The change in the peak current of the polymer is discussed by the following equation:

$$i_p = n^2 F^2 \Gamma v / 4RT$$

The linear relation between peak current intensities of polymer and scan rate is observed for non-diffusion controlled mechanism. The peak current is linearly dependent on the scan rate. A linear relationship was found between the peak current and the scan rate for **P-SBTTh** and **P-SBTPh**. This relationship indicates that the electroactive polymer films were well adhered and the redox processes were non-diffusion controlled.



**Figure 26.** Cyclic voltammograms of **P-SBTTh** in 0.1 M TBAPF<sub>6</sub>/ACN at scan rates of 100, 150, 200, 250 and 300 mV/s.



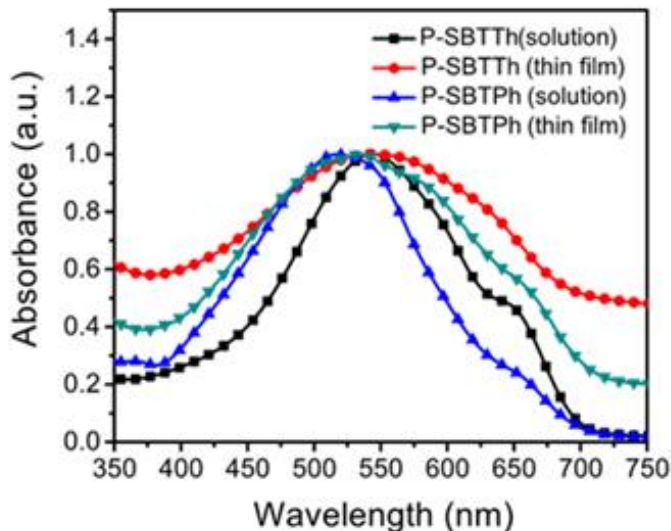
**Figure 27.** Cyclic voltammograms of **P-SBTPh** in 0.1 M TBAPF<sub>6</sub>/ACN at scan rates of 100, 150, 200, 250 and 300 mV

### 3.1.4 Optical Properties of P-SBTTh and P-SBTPh

Optical properties of polymers were investigated in thin form and in dilute chloroform solution and presented in **Figure 28**. All relevant data were summarized in **Table 3**. The maximum absorptions for **P-SBTTh** and **P-SBTPh** in solution were observed at 544 nm and 516 nm respectively. The absorption spectra of the polymer film of **P-SBTPh** (21nm) showed obvious red shift relative to the one recorded in solution. This phenomenon is the result of the molecular organization in the thin film to form more ordered structures and well-ordered intermolecular interaction in solid state. It is noteworthy to state that **P-SBTTh** did not show any meaningful red shift in thin film form, instead a slightly broadened absorption peak was observed. This can be ascribed to small reorganization in thin film and pre-aggregation in solution [70].

When optical properties of **P-SBTTh** and **P-SBTPh** were compared with their thiophene analogs, 23 nm and 21 nm red shifts were observed in maximum absorptions of **PTBTTh** and **PTBTPh** respectively. Incorporation of selenophene in polymer

backbone not only causes a red shift but also lowers the optical band gap compared to thiophene analogs. Differences as much as 0.21 eV and 0.23 eV in the optical band gaps of **PTBTTh** and **PTBTPh** were observed. These findings can be attributed to the shorter inter-ring C–C bonds of selenophene containing polymers compared to thiophene analogs. Therefore, the quinoid structure will make a significantly greater contribution to the ground state of **P-SBTTh** and **P-SBTPh** compared to **PTBTTh** and **PTBTPh** and energetically less stable quinoidal form can result in the reduction of band gap [59,60].



**Figure 28.** Normalized absorption spectra of **P-SBTTh** and **P-SBTPh** in chloroform solution and in thin film form

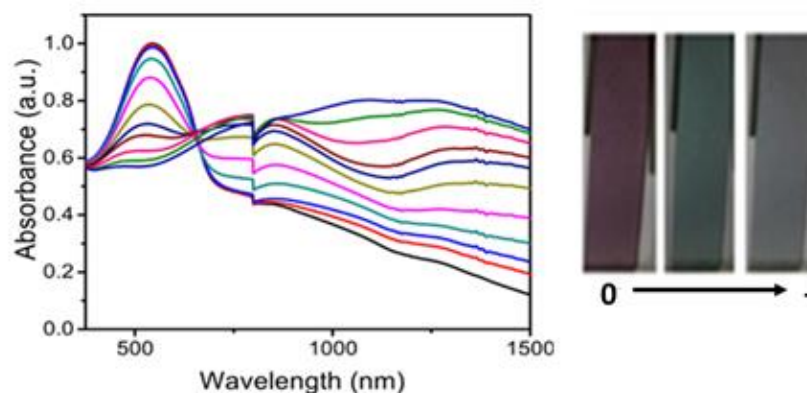
**Table 3.** Summary of optical properties of polymers

Optical Properties		
Polymer	Thin film $\lambda_{\max}$ (nm)	Solution $\lambda_{\max}$ (nm)
P-SBTTh	550	544
P-SBTPh	537	516

### 3.1.5 Spectroelectrochemical Properties of P-SBTTh

Spectroelectrochemistry studies were performed to gain a deeper idea on the changes in the optical properties during doping process. Therefore, polymer was spray coated and absorbance changes were studied by increasing the potential gradually from 0.0 V to 1.3 V for **P-SBTTh** in 0.1 M TBAPF<sub>6</sub>/ACN electrolyte-solvent couple by UV-Vis-NIR spectroscopy. As shown in **Figure 29**, **P-SBTTh** exhibited one absorption band arising from  $\pi$ - $\pi^*$  transition. **P-SBTTh** has a dominant red shifted absorption centered at 550 nm with an onset of 806 nm. The optical band gap ( $E_g^{\text{opt}}$ ) was calculated from onset of absorption of the thin film and estimated as 1.54 eV for **P-SBTTh**. **P-SBTTh** was oxidized in a stepwise manner, neutral state absorption started to decrease and new bands intensified in the NIR region at around 800 nm and 1280 nm due to formation of polarons and bipolarons. **P-SBTTh** exhibited purple color in its neutral states and stepwise oxidation caused the observation of different colors. Transmissive grey color was observed for **P-SBTTh** in their oxidized states. The absence of transparent color in bleached state can be attributed to the tailoring of polaron peaks in visible region.



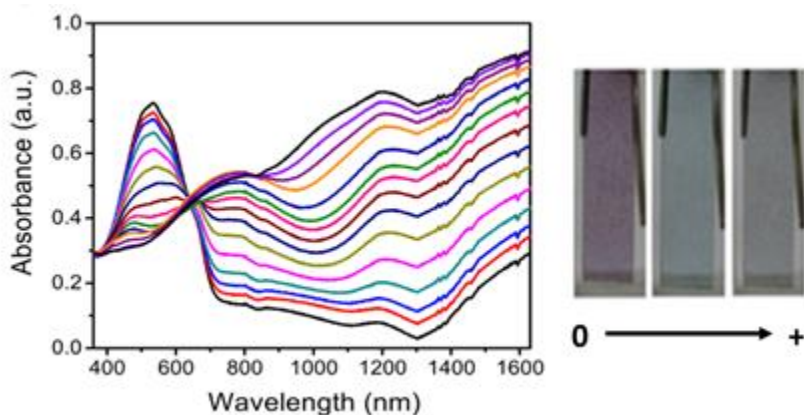


**Figure 29.** UV–Vis–NIR spectra and colors of **P-SBTTh** potentials between 0 and 1.3 V.

### 3.1.6 Spectroelectrochemical Properties of **P-SBTPh**

For the same purpose, spectroelectrochemistry studies were performed for the **P-SBTPh**. It was spray coated and absorbance changes were studied by increasing the potential gradually from 0.0 V to 1.35 V in 0.1 M TBAPF<sub>6</sub>/ACN electrolyte-solvent couple by UV-Vis-NIR spectroscopy. As shown in **Figure 30**, **P-SBTPh** exhibited one absorption band arising from  $\pi$ - $\pi^*$  transition. **P-SBTPh** revealed a maximum absorption band at 516 nm with an onset of 764 nm. The optical band gap of **P-SBTPh** was calculated from onset of absorption of the thin film and estimated as 1.62 eV. Even though, **P-SBTPh** had an extended  $\pi$ -conjugation length compared to **P-SBTTh**, it had larger band gap. This indicates that the bandgaps were predominantly determined by the electron-donating ability of the additional donor units (benzene < thiophene) in the polymer backbones. Then, **P-SBTPh** was stepwise oxidized, neutral state absorptions started to decrease and new bands were intensified in the NIR region at around 800 nm and 1280 nm for **P-SBTTh** and 770 nm and 1200 nm for **P-SBTPh** due to formation of polarons and bipolarons. Both polymers exhibited purple color in their neutral states and stepwise oxidation caused the observation of different colors.

Transmissive grey color was observed for both polymers in their oxidized states. The absence of transparent color in bleached state can be attributed to the tails of polaron peaks in the visible region.

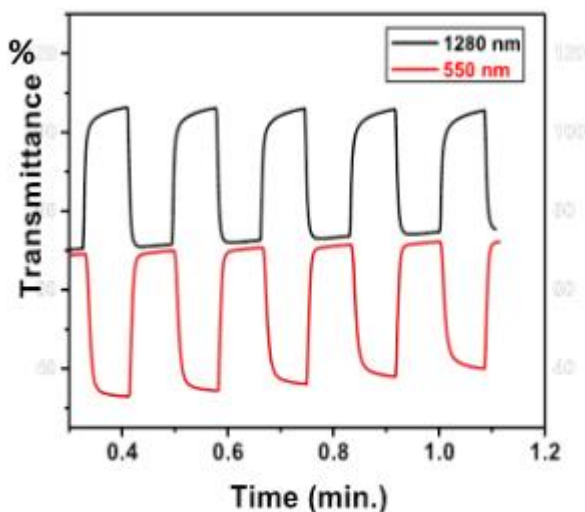


**Figure 30.** UV–Vis–NIR spectra and colors of **P-SBTPh** potentials between 0 and 1.35 V.

### 3.1.7 Kinetic Properties of P-SBTTh and P-SBTPh

In addition to optical, electrochemical and spectroelectrochemical properties, switching times and optical contrast are other important aspects for electrochromic device applications. Therefore, chronoamperometry studies of polymers were performed to calculate optical transmittance and switching times by stepping potentials between neutral and oxidized states with residence times of 1 s for **P-SBTTh** and 5 s for **P-SBTPh**. Optical contrast and switching times of the **P-SBTTh** in different wavelengths were summarized in **Table 4** and shown in **Figure 31**. **P-SBTTh** revealed 36 % and 37 % optical contrasts at 550 and 1280 nm. For **P-SBTPh**, optical properties and switching times were summarized in **Table 5** and shown in **Figure 32**. **P-SBTPh** revealed lower optical contrast values in visible and NIR regions compared to **P-SBTTh** and

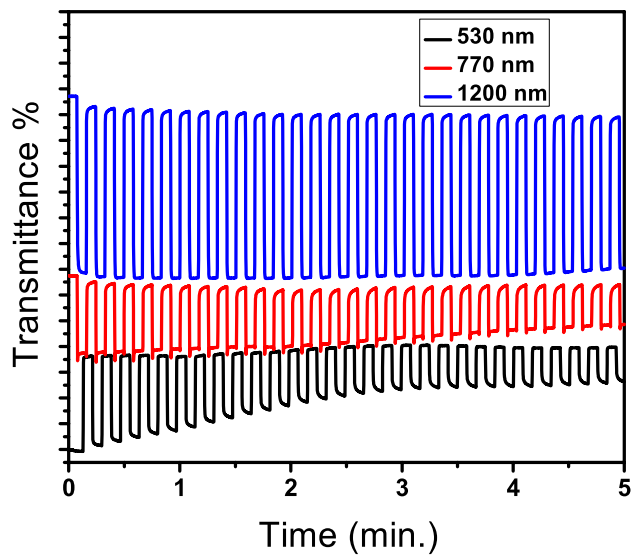
revealed 17 % in visible region (530 nm) and 14 % and 33 % in NIR region (770 and 1200 nm). Switching times were determined and calculated in different wavelengths for **P-SBTTh** and **P-SBTPh** and summarized in **Table 2**. Switching times of both polymers were less than 1 second. Calculated switching times of **P-SBTTh** were 0.6 s (550 nm) and 0.7 s (1280 nm). **P-SBTPh** showed fast switching times compared to **P-SBTTh** and switched between neutral and oxidized regimes in 0.2 s (530 nm), 0.4 s (770 nm), and 0.4 s (1200 nm).



**Figure 31.** Percent transmittance change and switching times of **P-SBTTh** in 0.1 M TBAFP<sub>6</sub>/ACN solution at its maximum wavelengths

**Table 4.** Optical contrast and switching times of the **P-SBTTh**

Polymer	Optical Contrast ( $\Delta T\%$ )	Switching time (s)
<b>P-SBTTh</b>	37 (1280 nm)	0.7
	36 (550 nm)	0.6



**Figure 32.** Percent transmittance change and switching times of **P-SBTPh** in 0.1 M TBAFP<sub>6</sub>/ACN solution at its maximum wavelengths

**Table 5.** Optical contrast and switching times of the **P-SBTPh**

Polymer	Optical Contrast ( $\Delta T\%$ )	Switching time (s)
<b>P-SBTPh</b>	33 (1200 nm)	0.4
	14 (770 nm)	0.4
	17 (530 nm)	0.2

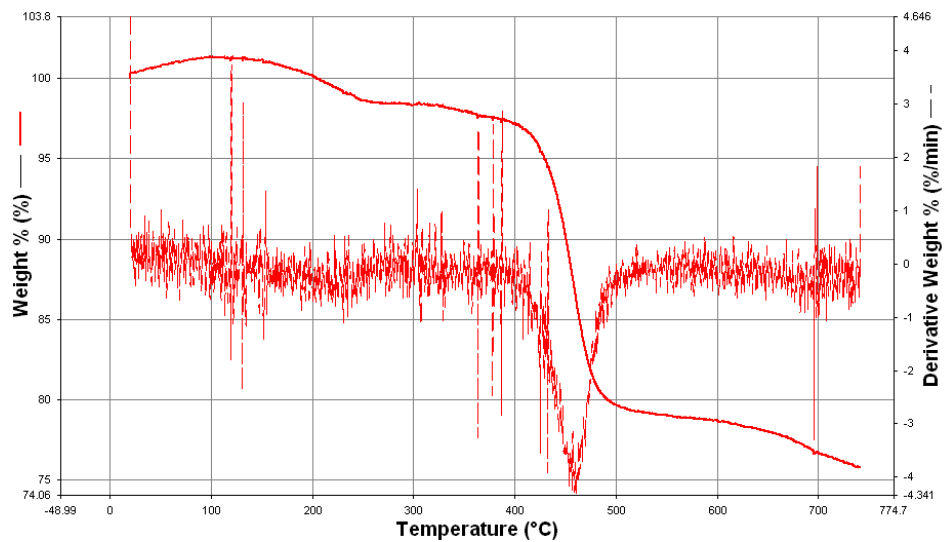
Both polymers exhibited moderate stability during kinetic studies. The reason for this can be explained by redox processes where polymers films are oxidized and reduced by insertion and extraction of ions in polymer film [69]. Therefore any remaining ions on polymer film can result in an irreversible redox process creating a stability problem. Although, incorporation of long alkyl chain can result in an increase in effective conjugation length and processability of polymer, conversely it may also hinder the exertion of counter ions from polymer backbone.

Although polymers showed moderate stability, they exhibited fascinating switching times. **P-SBTPh** showed outstanding switching times (0.2 s at 530 nm). To the best of our knowledge switching time of **P-SBTPh** is the fastest data measured up to now in all benzotriazol containing polymers. In addition to fast switching times, this polymer showed high stability at NIR region with reasonable optical contrast. By keeping in mind that the most electrochromic device (ECD) application needs fast switching times such as video displays (30-50 ms), information displays, e-readers, and point-of-purchase advertising (100-500 ms), so that **P-SBTPh** can be a reasonable candidate for NIR device applications in terms of fast switching time (0.4 s) with reasonable optical contrast.

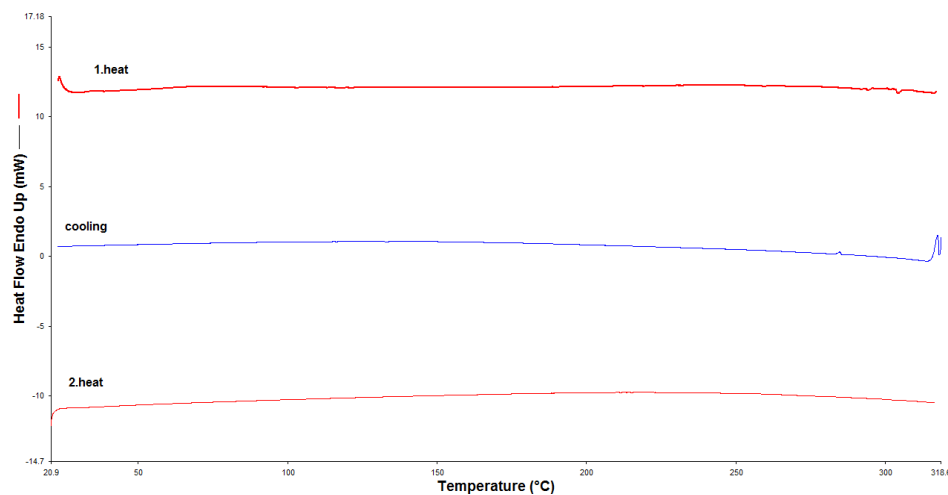
### 3.1.8 Thermal Analysis

Thermal behaviors of polymers were examined by thermogravimetry analyses (TGA) under nitrogen atmosphere with a heating rate of  $10^{\circ}\text{C min}^{-1}$ . 5% mass loss was observed at  $422^{\circ}\text{C}$  for **P-SBTTh** and 5 % weight loss was observed at  $370^{\circ}\text{C}$  for **P-SBTPh**. Differential scanning calorimetry (DSC) was utilized to characterize the

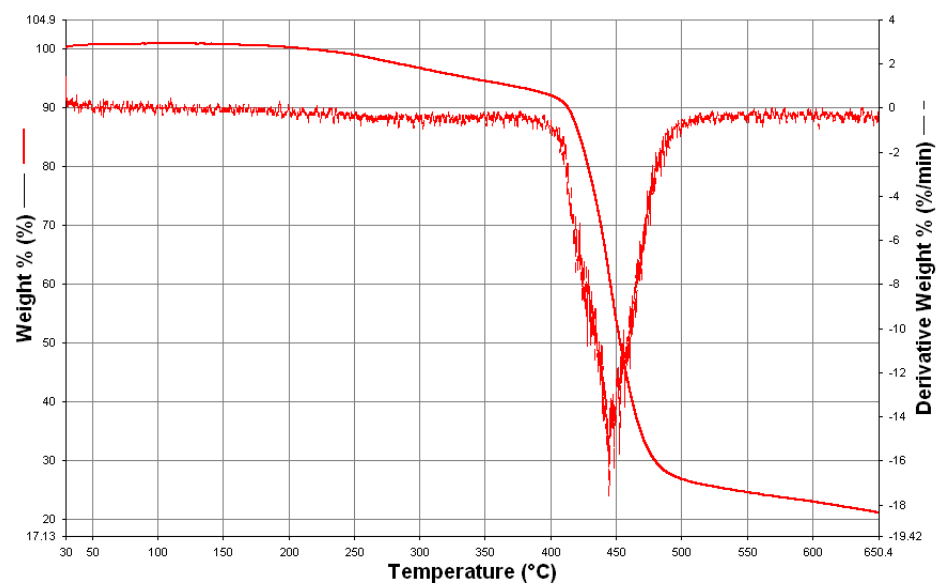
thermal transitions in the polymers and **P-SBTTh** did not show any phase transition. **P-SBTPh** revealed a glass transition-like feature at 143 °C.



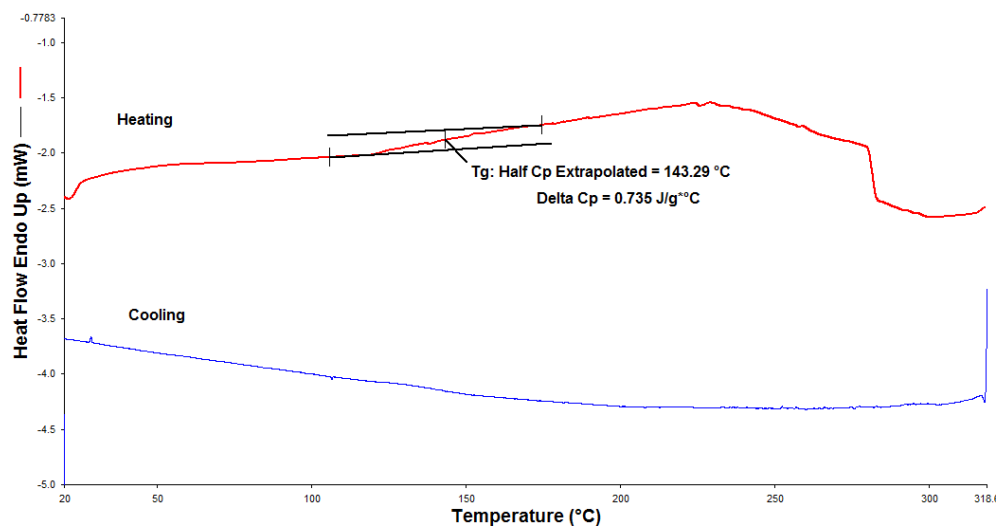
**Figure 33.** TGA result of **P-SBTTh**



**Figure 34.** DSC result of P-SBTTh



**Figure 35.** TGA result for P-SBTPh



**Figure 36.** DSC result of **P-SBTPh**



## CHAPTER 4

### CONCLUSION

Two novel polymers containing selenophene as a  $\pi$ -bridge were synthesized via Stille and Suzuki coupling reactions. Selenophene incorporation in polymer backbone resulted in several changes in optical and electronic properties of BTz based polymer. It caused low band gap and red shifted absorption spectra compared to their thiophene analogs. Both polymers exhibited multichromism switching between purple, blue and highly transmissive grey color. Kinetic studies indicated that both polymers showed moderate stability but they exhibited a fascinating property in terms of switching times. Especially, switching time of **P-SBTPh** at UV-vis region is the fastest data reported up to now in all benzotriazol based copolymers. In addition to fast switching times, this polymer showed high stability at NIR region with reasonable optical contrast (33%). Therefore, structural modification, done by insertion of different chalcogenophene rings in the polymer backbone, seems a reasonable way to control optical, electrochemical and electrochromic properties of BTz based polymers. Therefore, further investigation of selenophene containing materials is needed to. From this perspective, design and synthesis of new materials are crucial for their use in different applications.



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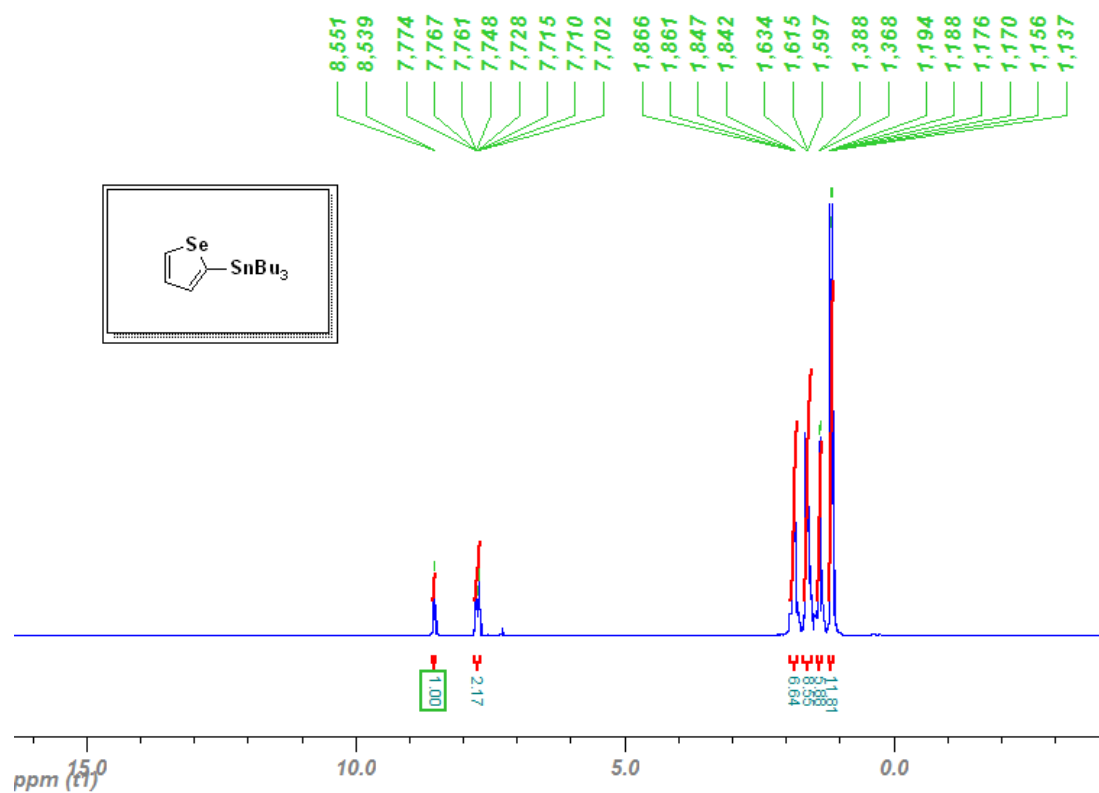


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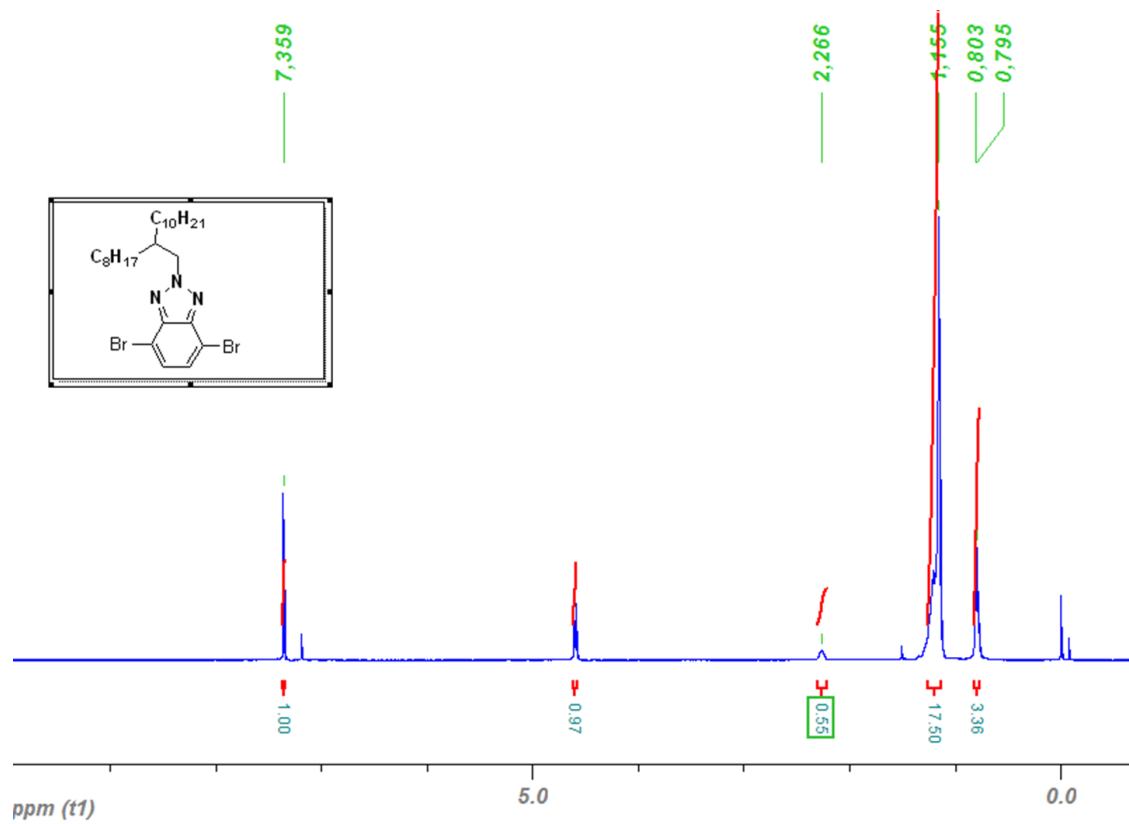


## APPENDIX A

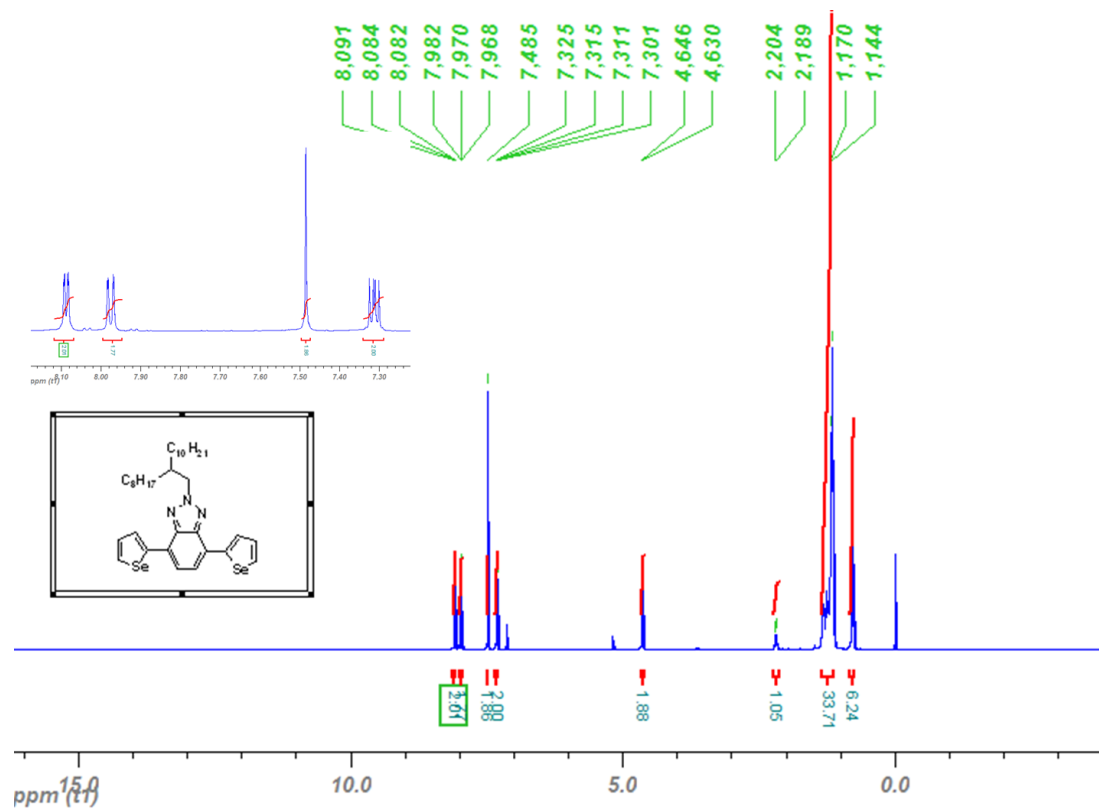
### NMR DATA



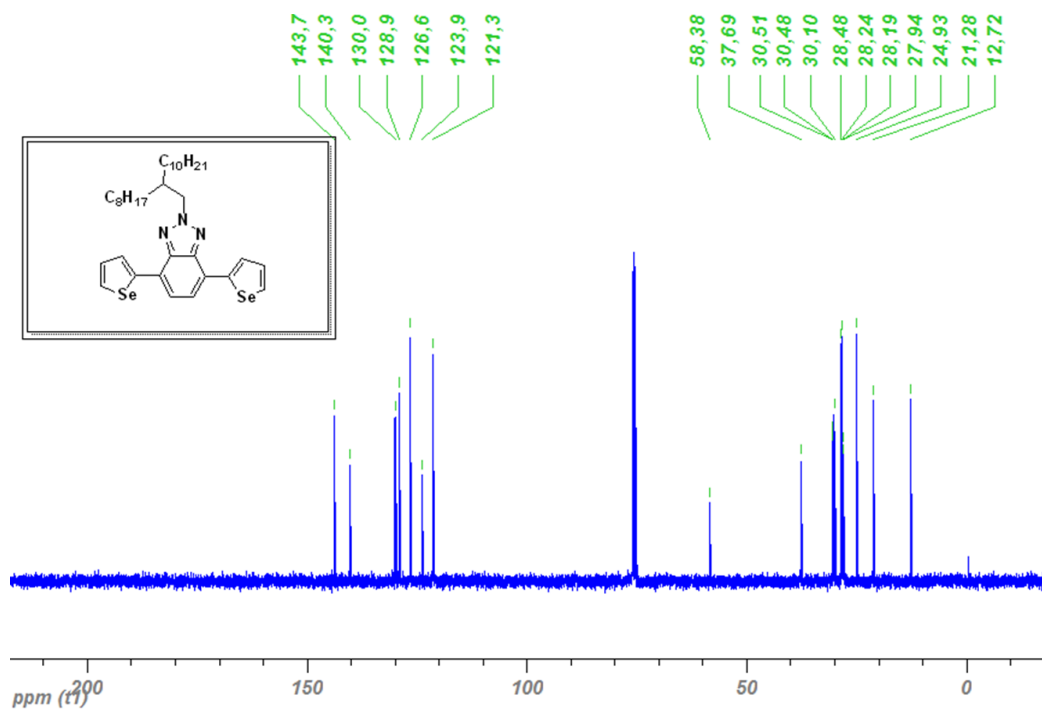
**Figure 37.**  $^1\text{H}$  NMR of Tributyl(selenophen-2-yl)stannane



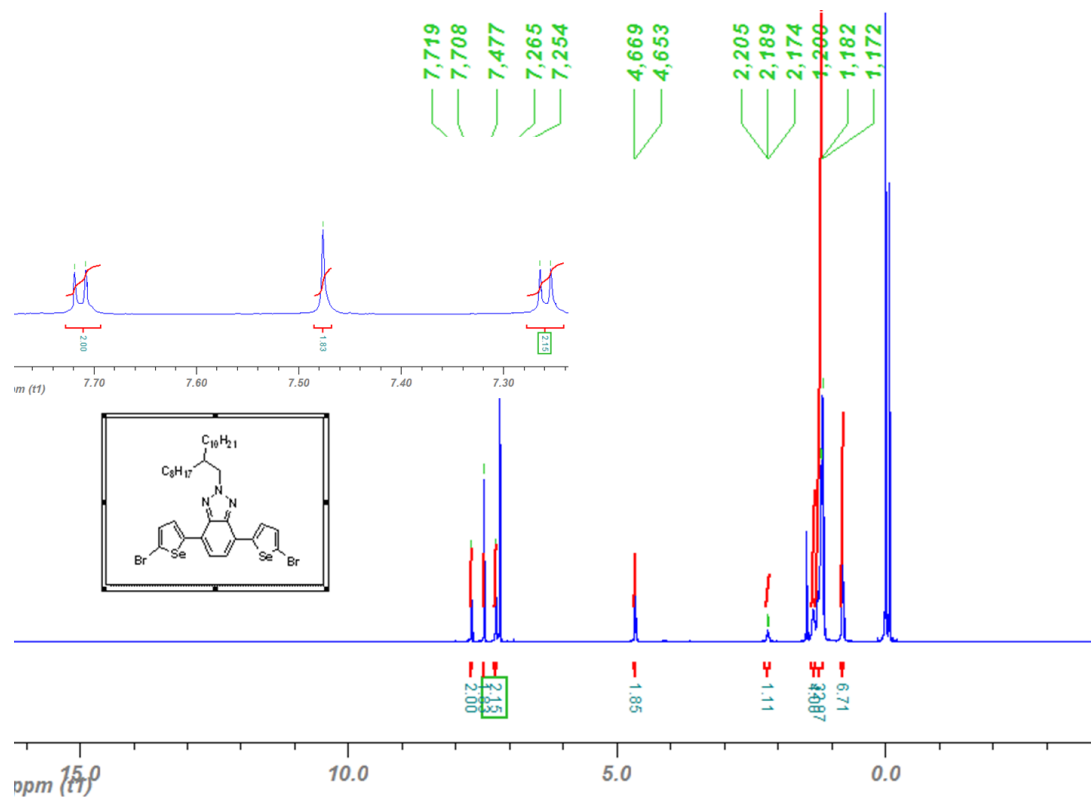
**Figure 38.**  $^1\text{H}$  NMR of 4,7-dibromo-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole



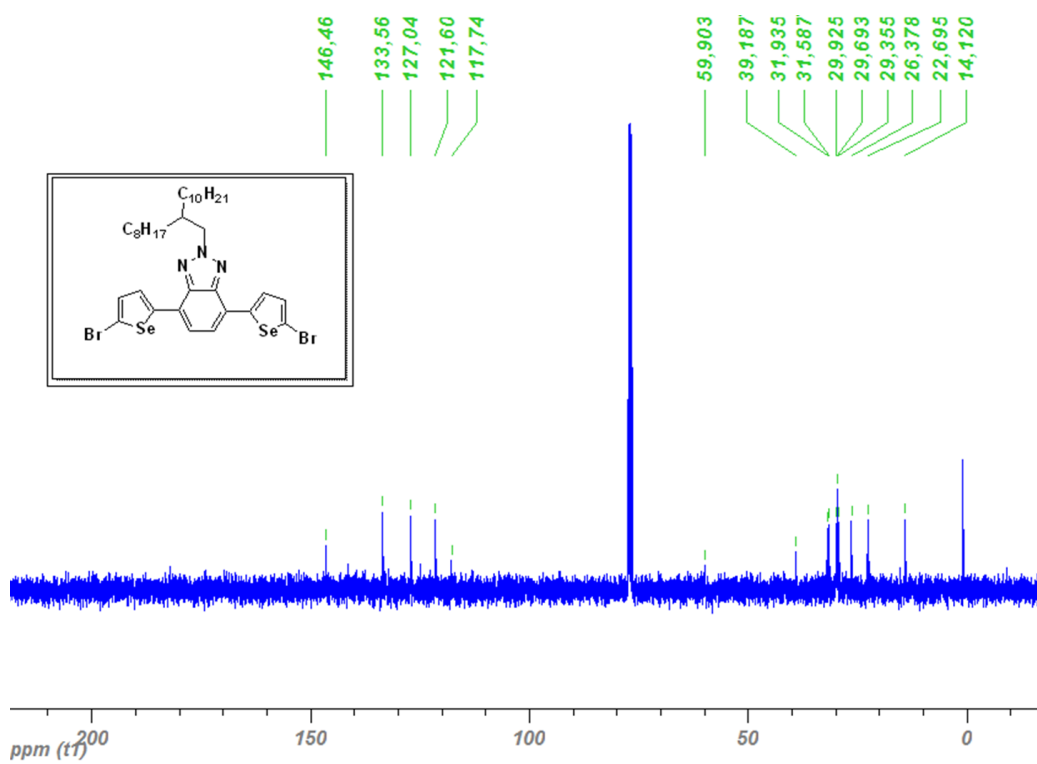
**Figure 39.**  $^1\text{H}$  NMR of 2-(2-octyldodecyl)-4,7-di(selenophen-2-yl)-2H-benzo[d][1,2,3]triazole



**Figure 40.**  $^{13}\text{C}$  NMR of 2-(2-octyldodecyl)-4,7-di(selenophen-2-yl)-2H-benzo[d][1,2,3]triazole

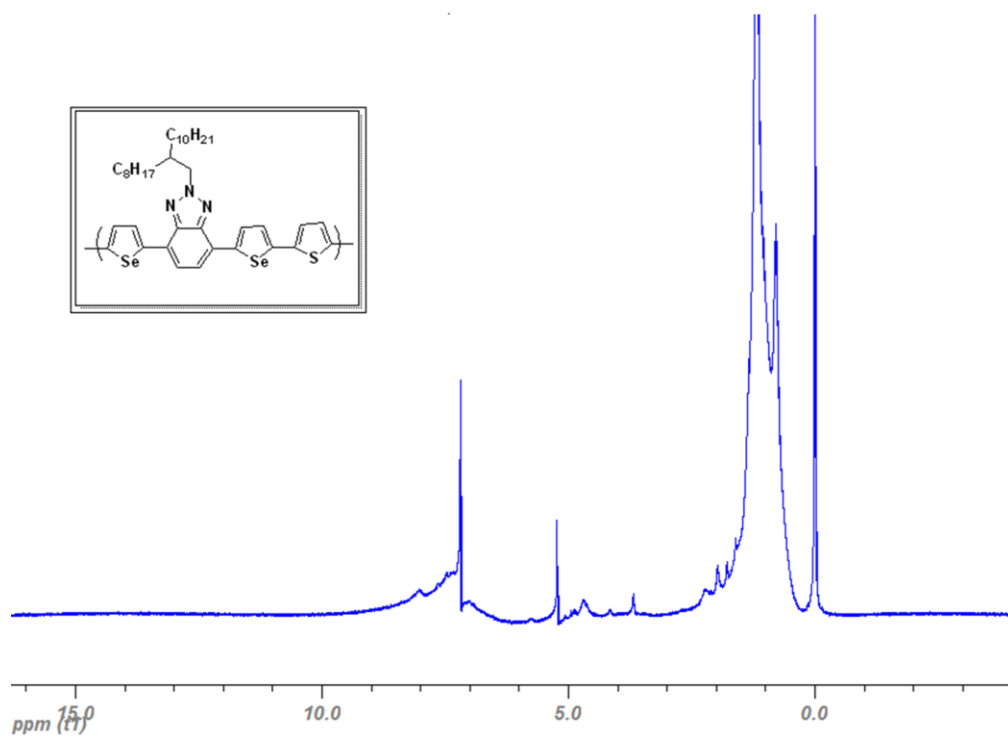


**Figure 41.**  $^1\text{H}$  NMR of 4,7-bis(5-bromoselenophen-2-yl)-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole

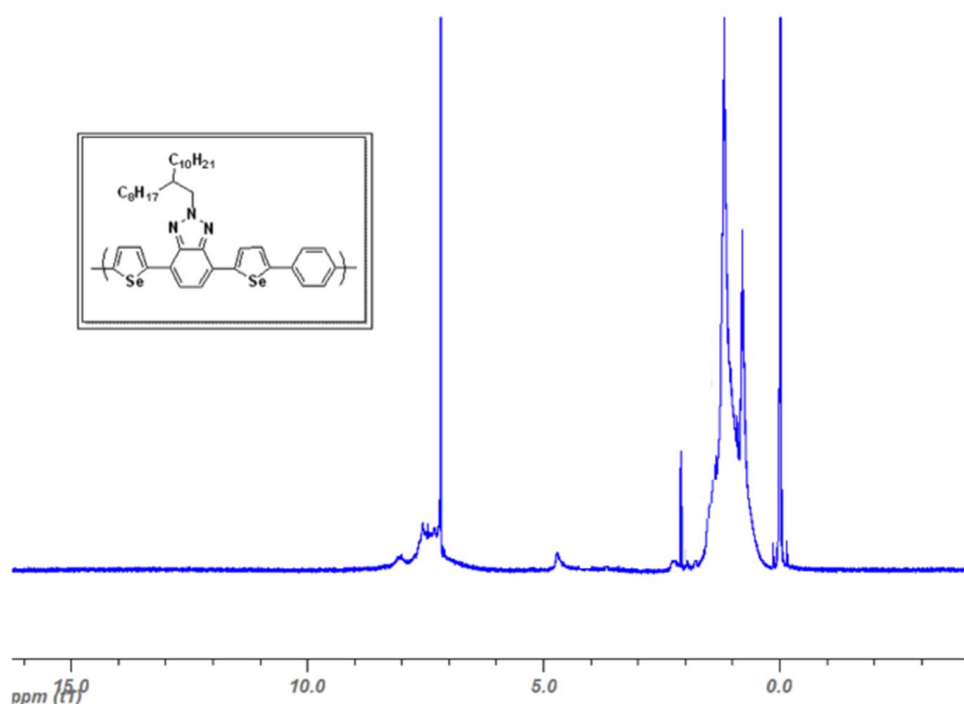


**Figure 42.** <sup>13</sup>C NMR of 4,7-bis(5-bromoselenophen-2-yl)-2-(2-octyldodecyl)-2H-benzo[d][1,2,3]triazole





**Figure 43.**  $^1\text{H}$  NMR of P-SBTTh



**Figure 44.**  $^1\text{H}$  NMR of P-SBTPh